

substance: Ti_2O_3

property: properties of doped Ti_2O_3

$\text{Ti}_2\text{O}_3\text{:V}$: Addition of small amounts of V appears to mimic the temperature increase. Studies of lattice parameters: [74R, 76C, 77R2, 77R3], axial and equatorial interatomic distances: Fig. 1. Lattice properties show marked changes [75B, 74S]. Raman spectrum shows the softening of the lowest A_{1g} and E_g modes (Fig. 2) to follow closely the behaviour of the pure material on heating (Fig. 3). The temperature variation of the A_{1g} and E_g modes is essentially suppressed for $x > 0.12$ in $(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$ (Fig. 4). The heat capacity anomaly (see below) at ca. 470 K is also suppressed for $x > 0.04$ [73B].

The heat capacity of V-doped Ti_2O_3 shows a marked anomaly at low temperature [73S2, 79M]. Initially this was interpreted in terms of a one dimensional conductivity model [72S]; but this idea was shown to be untenable [73Z1, 73Z2], and an electronic model invoking a very narrow V a_1 band was proposed. Subsequent magnetic measurements [75D1, 75D2, 75E, 77D, 78D, 79D1] have established that, for $x < 0.02$, localized magnetic moments exist on V, but decrease with increasing x . Above $x = 0.007$ a spin-glass forms at very low temperature, and it is this transition from spin-glass to a paramagnetic metal with no magnetic moment on the V that gives rise to the heat capacity anomaly [79D1].

Conductivity in $\text{Ti}_2\text{O}_3\text{:V}$: Fig. 5. The transition onset temperature remains constant up to the point where it disappears, and the apparent activation energy falls smoothly from 0.0254 eV in pure Ti_2O_3 to zero at 2.6 at % V [70C]. Similar Seebeck data: Fig. 6. Phase diagram: Fig. 7 and [80D].

$\text{Ti}_2\text{O}_3\text{:Sc}$: Structural changes reported in [77R1], electrical properties in [74C]. ρ first decreases with x and then increases. Seebeck coefficient shows much reduced phonon effect at low temperatures, perhaps because impurity scattering dominates.

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Fig. 1.

$(\text{Ti}_{1-x}\text{M}_x)_2\text{O}_3$, $\text{M} = \text{V}, \text{Sc}, \text{Al}$. Interatomic distances vs. composition parameter x [77R2].

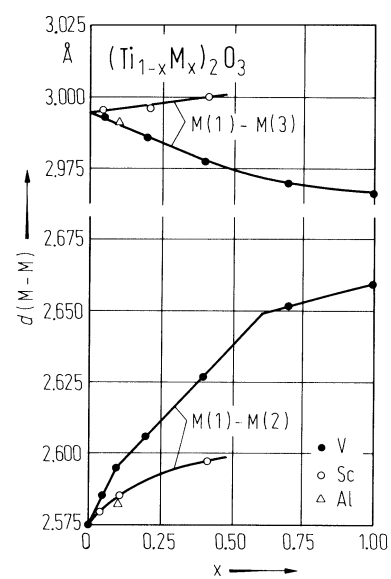


Fig. 2.

$(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$. Phonon wavenumbers vs. composition parameter x at 300 K [74S].

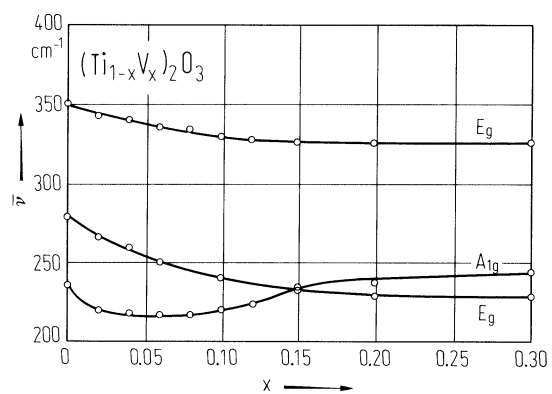


Fig. 3.

Ti_2O_3 . Wavenumbers of Raman frequencies vs. temperature [74S].

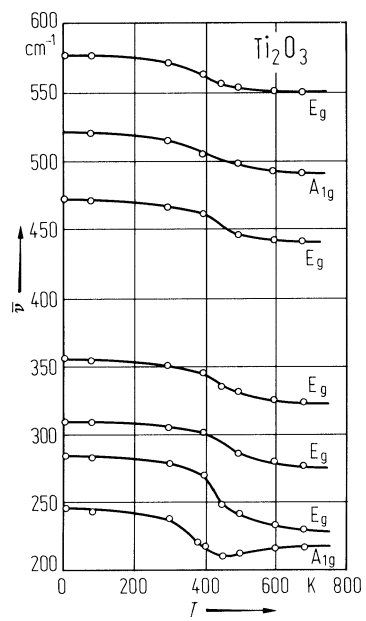
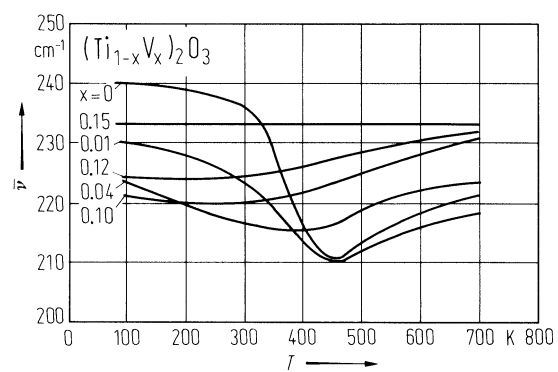


Fig. 4.

$(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$. Wavenumbers of the A_{1g} -mode vs. temperature for samples of different compositions [74S].



(Ti_{1-x}V_x)₂O₃. Resistivity vs. reciprocal temperature for polycrystalline samples of different compositions [70C]. Dashed line: transition temperature from semiconductor to metal.

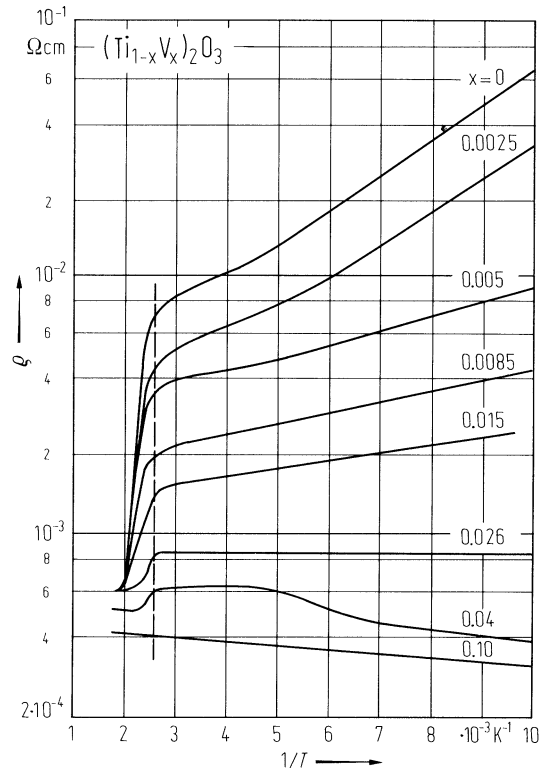


Fig. 6.

$(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$. Seebeck coefficient vs. temperature for polycrystalline doped samples of different compositions and for single crystals of Ti_2O_3 [73S1].

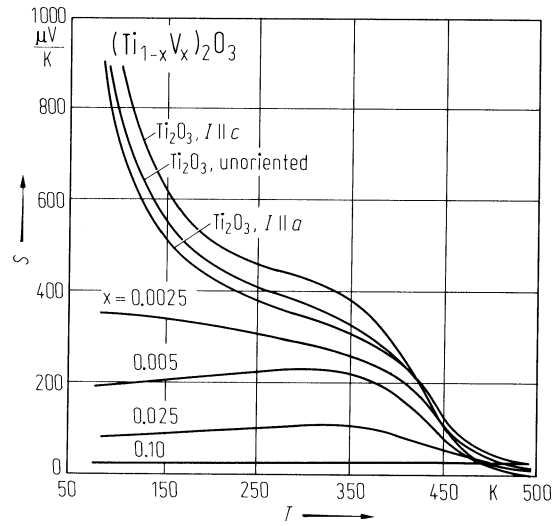


Fig. 7.

$(\text{Ti}_{1-x}\text{V}_x)_2\text{O}_3$. Phase diagram. A: spin glass, B: micromagnetic, C: antiferromagnetic metal, D: antiferromagnetic semiconductor, E: paramagnetic semiconductor, F: paramagnetic metal [79D2].

