

**substance: PtSb<sub>2</sub>**

**property: transport properties**

See also document .

**electrical resistivity**

(Figs. 2...5, see also Fig. 1 and 22)

$\rho_i$	1 $\Omega$ cm	$T = 70$ K	intrinsic value	68R,
	0.1 $\Omega$ cm	$T = 90$ K		83D
	0.01 $\Omega$ cm	$T = 135$ K		
	0.001 $\Omega$ cm	$T = 250$ K		

**Hall coefficients**

(Figs. 5...8)

$R_H$	+ 4 cm <sup>3</sup> C <sup>-1</sup>	$T = 2...120$ K, $B < 3$ T	single crystal, intrinsic near RT	73A
	+ 3.0 cm <sup>3</sup> C <sup>-1</sup>	$T = 77...120$ K, $B < 1.1$ T	Te doped single crystal with $\rho = 0.0035$ $\Omega$ cm at 77 K	65D
	+ 103 cm <sup>3</sup> C <sup>-1</sup>	$T = 77$ K	single crystal with $\rho = 0.0176$ $\Omega$ cm at 77 K	83D
	- 110 cm <sup>3</sup> C <sup>-1</sup>	$T = 77$ K	single crystal with $\rho = 0.58$ $\Omega$ cm at 77 K	68R
	- 600 cm <sup>3</sup> C <sup>-1</sup>	$T = 62.5$ K		

**thermoelectric powers**

(Figs. 9 and 10)

$S$	+ 80 $\mu$ V K <sup>-1</sup>	$T = 300$ K	polycrystalline sample	63H
	+ 180 $\mu$ V K <sup>-1</sup>	$T = 80$ K	pure and Au doped (n-type PtSb <sub>2</sub> by doping with Te)	65J
	- 200 $\mu$ V K <sup>-1</sup>		Te doped single crystal with	83D
	- 825 $\mu$ V K <sup>-1</sup>	$T = 25$ K	donor excess $n_d - n_a \approx 10^{17}$ cm <sup>-3</sup>	
	0 $\mu$ V K <sup>-1</sup>	$T = 100$ K	and $n_a/n_d \approx 0.8$ ; $\Delta T \leq 0.2$ K	
	+ 150 $\mu$ V K <sup>-1</sup>	$T = 125$ K		

**mobility of charge carriers**

(in cm<sup>2</sup>/V s)

$\mu_n$	420	$T = 300$ K	from $\rho(T)$ and $R_H(T)$ in the mixed- conduction range;	68R
	3700	$T = 77$ K	$\mu_n = 2.07 \cdot 10^6 T^{-3/2} \exp(12.6 \text{ K}/T)$	
	2100	$T = 50$ K	maximum of $R_H \sigma$	68E
	$3 \cdot 10^6 T^{-3/2}$	$T = 7...50$ K	acoustic-phonon-limited mobility, from $\rho(T)$ and $R_H(T)$	83D
$\mu_p$	830	$T = 300$ K	$\mu_n = 4.15 \cdot 10^6 T^{-3/2} \exp(12.6 \text{ K}/T)$	68R
	7300	$T = 77$ K	$\approx 6.55 \cdot 10^6 T^{-1.57}$ above 100 K	
	$7.5 \cdot 10^6 T^{-3/2}$	$T = 7...50$ K		83D

temperature dependence: Figs. 11...13

In the impurity-conduction range the mobility is temperature independent which suggests that a metallic rather than a hopping type of impurity conduction is dominant [68R]. Between 10 and 40 K the mobilities are limited primarily by ionized-impurity scattering [68R]. The occurrence of the near  $T^{-3/2}$  dependence of  $\mu_n$  and  $\mu_p$  over a wide temperature range shows that acoustic-mode lattice scattering limits the free-carrier mobilities in sufficiently pure PtSb<sub>2</sub> [68R, 68E]. Better agreement with experimental data by taking into account scattering by ionized impurities at low  $T$  and by optical phonons at higher  $T$  [83D]. Hole and electron mobility are independent of the electric field up to the impact ionization threshold, 1 and 3 kV cm<sup>-1</sup>, respectively [80D].

**mobility ratio**

(Fig. 11)

$b$	0.57	$T \approx 110$ K	from $R_H^{\max}/R_H^{\text{exh}} = (1-b)^2/4b$ , taking $R_H^{\max}$ at 110 K and $R_H^{\text{exh}}$ in the exhaustion region at 74 K	68E
	0.5	$T = 78 \dots 200$ K	same procedure as above. Te doped single crystals with $n_d - n_a = 0.3 \dots 3.2 \cdot 10^{16} \text{ cm}^{-3}$	76A
	0.47...0.52	$T = 80 \dots 300$ K	from $R_H$ in the acoustic phonon scattering region of different samples	68R
	0.4	$T = 240$ K	from $R_H(T)$ and $S(T)$	73A
	1	$T = 560$ K	sign change of $R_H(T)$	
	0.4	$T = 10 \dots 100$ K	acoustic phonon scattering only; total $b$ increases with $T$	83D
<b>intrinsic carrier concentration</b>				
$n_i$	$7.7 \cdot 10^{18} \text{ cm}^{-3}$	$T = 300$ K	calculated from the experimental value of $\rho_i$ and the equations given above for $\mu_n$ and $\mu_p$	68R
	$2.2 \cdot 10^{15} \text{ cm}^{-3}$	$T = 77$ K	calculated as $n_i = 2.72 A \exp(-0.11 \text{ eV}/2kT)$ with $A = 4.84 \cdot 10^{15} T^{3/2}$ [ $K^{-3/2} \text{ cm}^{-3}$ ]	
	$1.3 \cdot 10^{16} T^{3/2} \exp(-640/T) [\text{cm}^{-3}]$	$> 100$ K	from $\rho(T)$ and $R_H(T)$ using a single parabolic conduction and valence band model	83D

For temperature dependence of  $n_i/T^{3/2}$ , see Figs. 14 and 15.

**magnetoresistance** in p-type crystals of low carrier concentration: Fig. 16.

oscillatory magnetoresistance in crystals of high carrier concentration: Figs. 17...19.

dependence of the **resistivity** on hydrostatic pressure: Fig. 20.

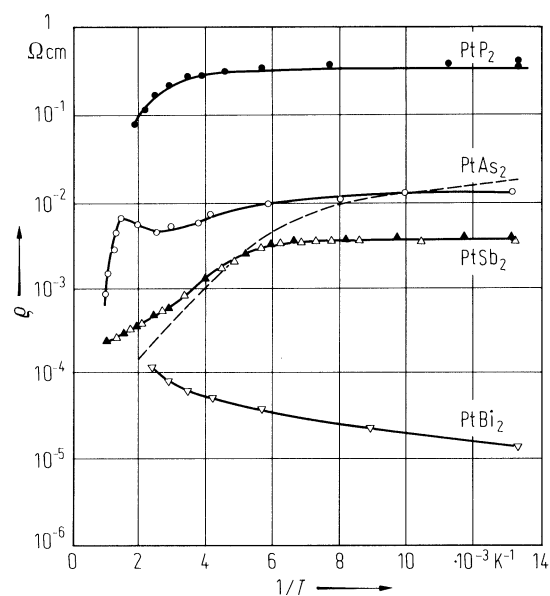
**piezoresistance** coefficients in the range 80...300 K: Fig. 21.

## References:

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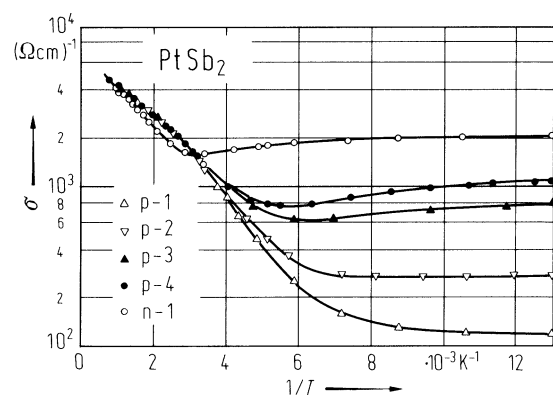
**Fig. 1.**

PtP<sub>2</sub>, PtAs<sub>2</sub>, PtSb<sub>2</sub>, PtBi<sub>2</sub>. Resistivity vs. reciprocal temperature [65J]. The data for semimetallic pyrite-type PtBi<sub>2</sub> (h) are added for comparison. PtP<sub>2</sub>: hot-pressed sample; PtAs<sub>2</sub> and PtBi<sub>2</sub>: sintered samples; PtSb<sub>2</sub>: open triangles - polycrystalline sample, full triangles - single crystals. The broken curve shows the measurements of [66B] on a sintered PtAs<sub>2</sub> sample.



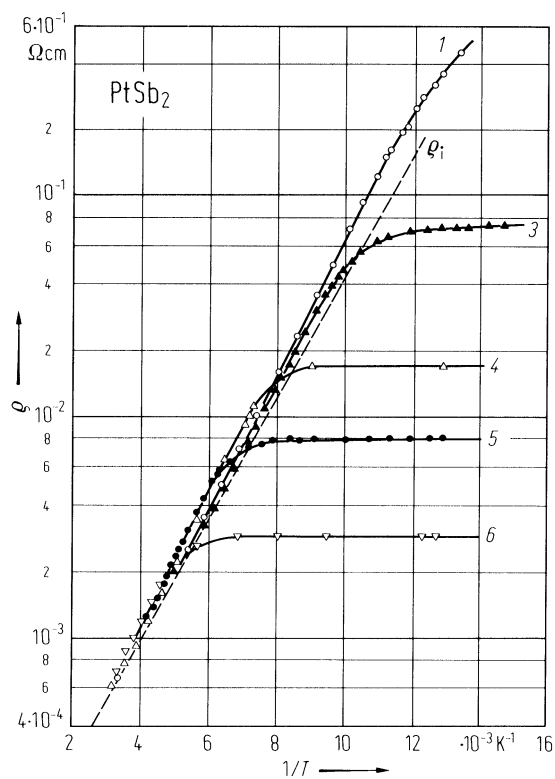
**Fig. 2.**

PtSb<sub>2</sub>. Conductivity vs. reciprocal temperature [65D]. Single crystal specimens as in Fig. 6.



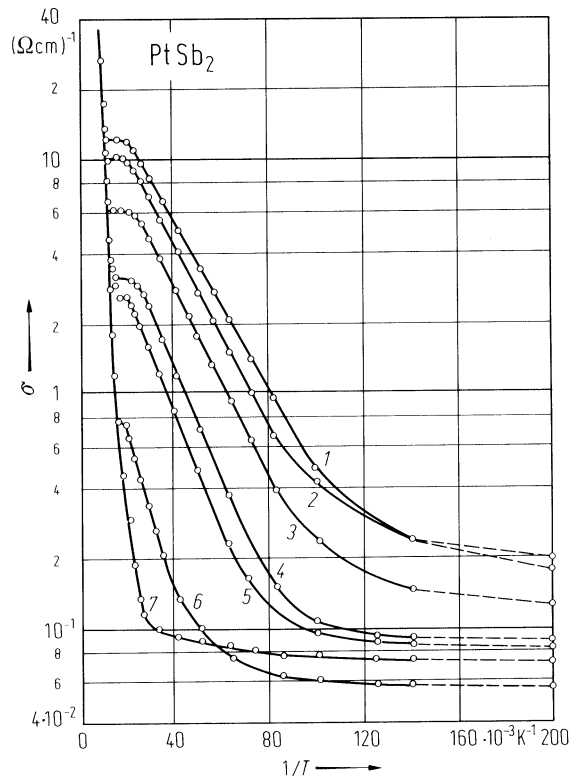
**Fig. 3.**

PtSb<sub>2</sub>. Resistivities vs. reciprocal temperature for n-type samples [68R]. Carrier concentrations given in Fig. 7.  
 $\rho_i$ : intrinsic resistivity.



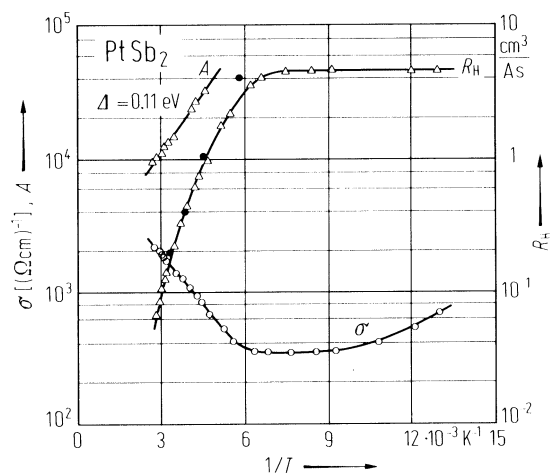
**Fig. 4.**

PtSb<sub>2</sub>. Electrical conductivity vs. reciprocal temperature [76A]. The specimens were fragments of a Te-doped single crystal with a Te concentration close to the concentration of residual acceptor impurities in "undoped n-type PtSb<sub>2</sub>". The donor (Te) concentration was approximately constant ( $n_d = 4 \cdot 10^{16} \text{ cm}^{-3}$ ) along the ingot and only the acceptor concentration varied. The samples were cut from the n-type part at right angles to the growth direction [110]. The sample numbers increase with the degree of compensation (as estimated from  $R_H$  in the depletion region).  $n_d - n_a$  in  $10^{16} \text{ cm}^{-3}$ : 1: 3.2, 2: 1.9, 3: 1.4, 4: 1.2, 5: 1.0, 6: 0.7, 7: 0.3.



**Fig. 5.**

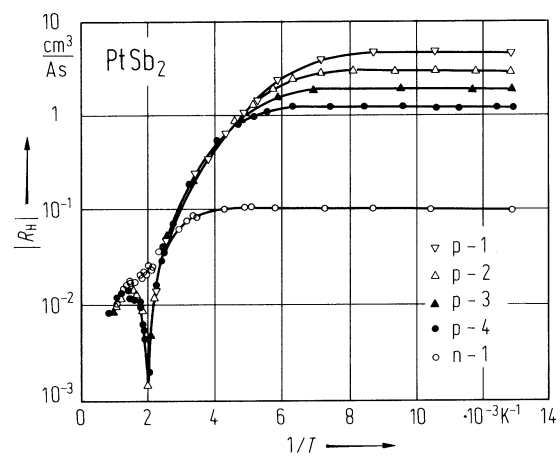
PtSb<sub>2</sub>. Electrical conductivity and Hall coefficient vs. reciprocal temperature for a p-type single crystal [73A]. The energy gap is derived from  $A = R_H T^{3/2} (m_n m_p / m_0^2)^{3/4} ((1+b)/(1-b))$ , where the density of states masses  $m_{n,p} = n_{C,V}^{2/3} m_{||}^{1/3} m_{\perp}^{2/3}$ , with  $n_{C,V}$  = number of energy ellipsoids,  $m_{||}$  and  $m_{\perp}$  = longitudinal and transverse effective masses, respectively.  $A$  in cm<sup>3</sup> K<sup>3/2</sup> As<sup>-1</sup>.





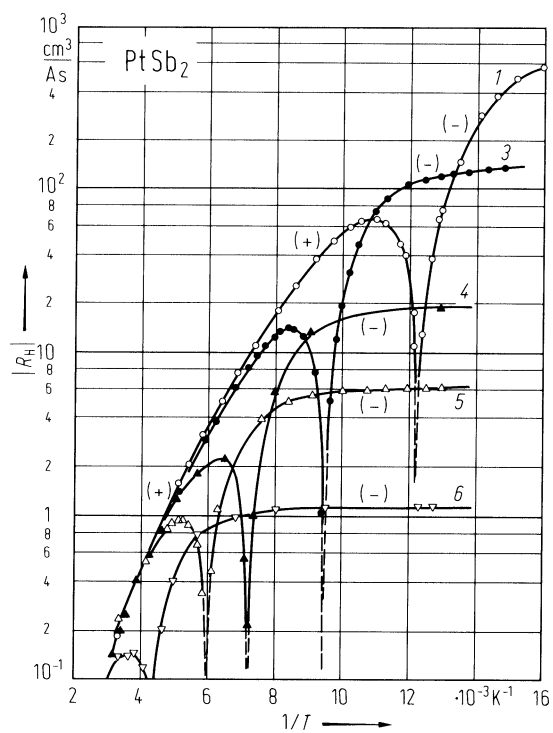
**Fig. 6.**

PtSb<sub>2</sub>. Absolute values of the Hall coefficient vs. reciprocal temperature of single crystals [65D].



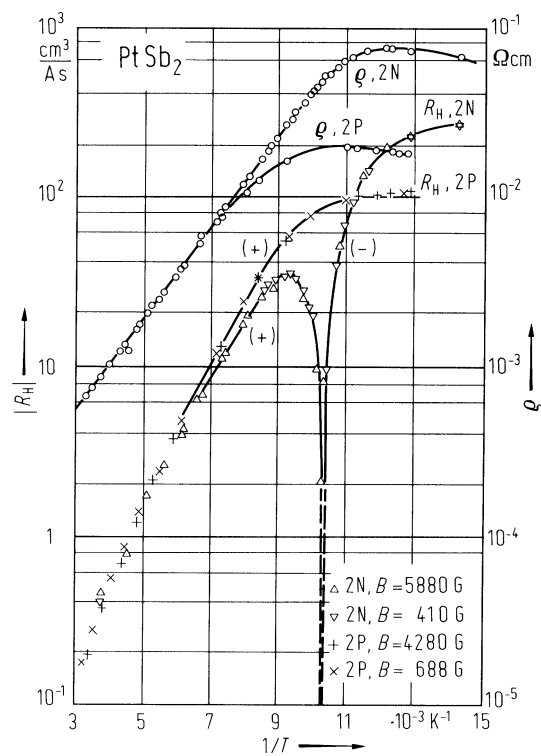
**Fig. 7.**

PtSb<sub>2</sub>. Hall coefficients vs. reciprocal temperature for n-type samples [68R]. Extrinsic carrier concentrations  $N_{\text{extr.}} = n - p = 1/eR_H$  (77K); 1:  $5.68 \cdot 10^{15} \text{ cm}^{-3}$ , 3:  $5.00 \cdot 10^{16} \text{ cm}^{-3}$ , 4:  $3.30 \cdot 10^{17} \text{ cm}^{-3}$ , 5:  $9.93 \cdot 10^{17} \text{ cm}^{-3}$ , 6:  $5.48 \cdot 10^{18} \text{ cm}^{-3}$ .



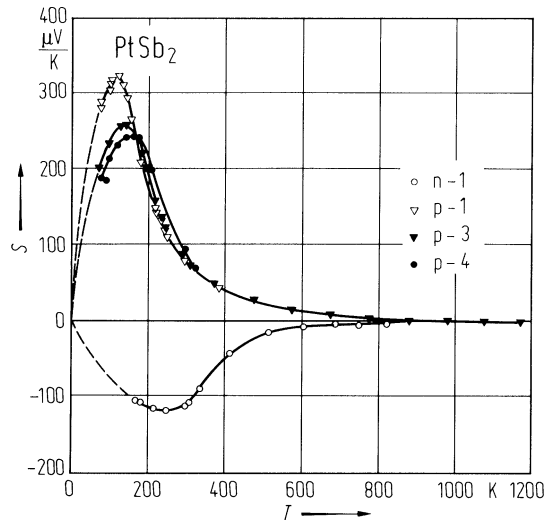
**Fig. 8.**

PtSb<sub>2</sub>. Hall coefficient and resistivity vs. reciprocal temperature in the mixed-conduction range for an n-type sample of low impurity contents [68R]. 2N:  $N_{\text{extr.}} = n - p \approx 2.4 \cdot 10^{16} \text{ cm}^{-3}$ ,  $\mu_{\text{H}}(77\text{K}) = 3140 \text{ cm}^2/\text{Vs}$ ; 2P:  $P_{\text{extr.}} = p - n \approx 6.06 \cdot 10^{16} \text{ cm}^{-3}$ ,  $\mu_{\text{H}}(77\text{K}) = 5840 \text{ cm}^2/\text{Vs}$ .



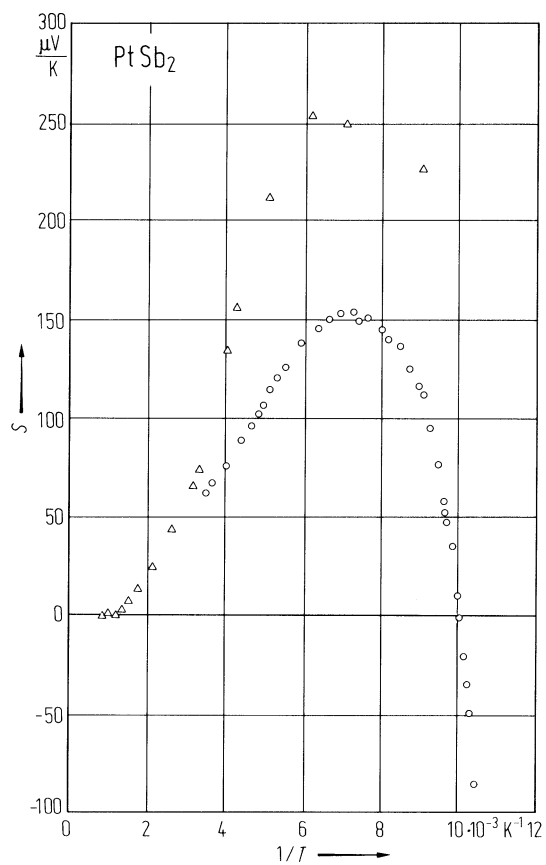
**Fig. 9.**

PtSb<sub>2</sub>. Seebeck coefficient vs. temperature [65D]. All samples were single crystals. See Fig. 6. Above  $\approx 1000$  K the Seebeck coefficient of sample p-3 is negative ( $\approx -1$   $\mu\text{V/K}$ ).



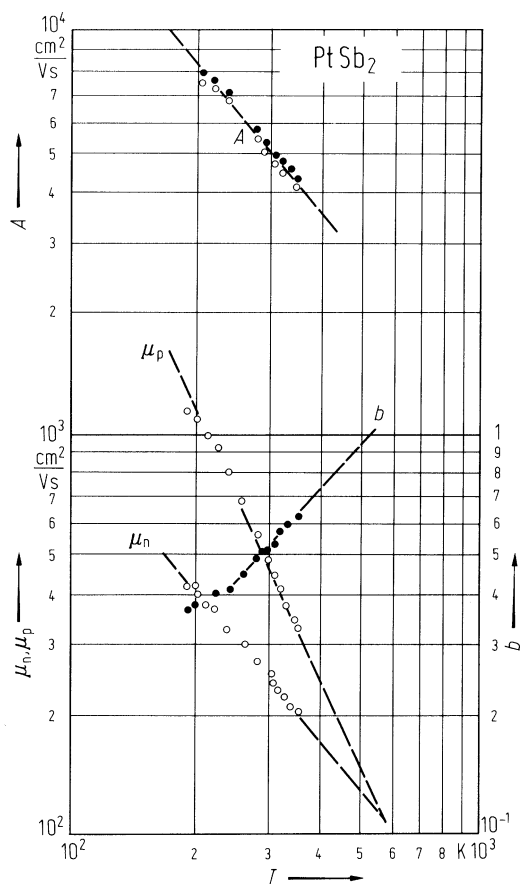
**Fig. 10.**

PtSb<sub>2</sub>. Seebeck coefficient vs. reciprocal temperature in the intrinsic region. Circles: Te doped single crystal with  $n_d - n_a \approx 10^{17} \text{ cm}^{-3}$ ,  $n_a/n_d \approx 0.8$  and  $(-eR_H)^{-1} = 6.6 \cdot 10^{16} \text{ cm}^{-3}$  [83D]. Triangles: Sample p-3 of [65D]; compare Fig. 6.



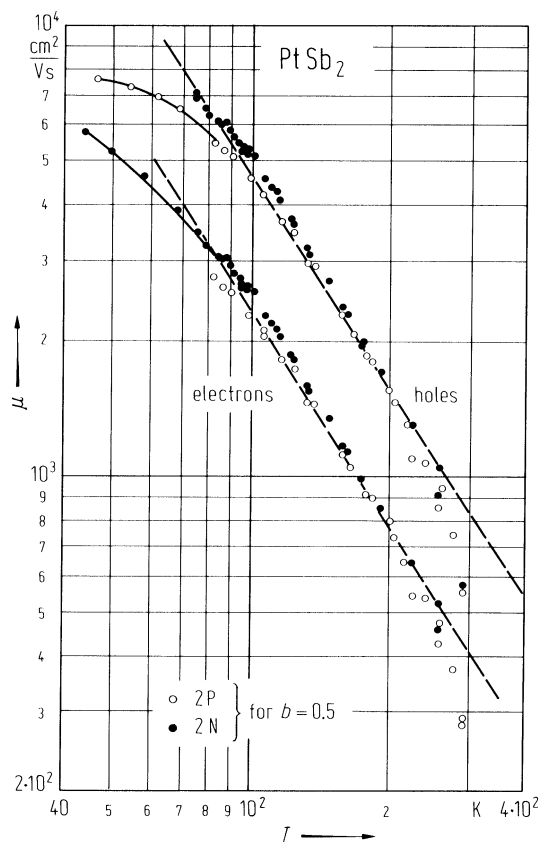
**Fig. 11.**

PtSb<sub>2</sub>. Mobilities of electrons and holes and mobility ratio  $b = \mu_n/\mu_p$  vs. temperature in the intrinsic region;  $b$  derived from  $S(T)$  [73A].  $A = \mu_p(m_p/m_0)^{5/2}$  (open circles) and  $\mu_n(m_n/m_0)^{5/2}$  (full circles) varies as  $\propto T^{-1.3}$ .



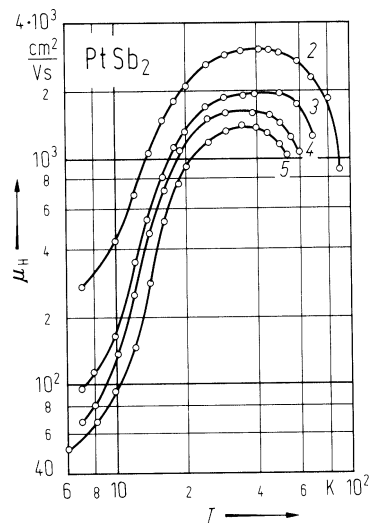
**Fig. 12.**

PtSb<sub>2</sub>. Electron and hole mobilities vs. temperature [68R]. The broken lines represent values calculated for the intrinsic range:  $\mu_p = 4.15 \cdot 10^6 T^{-3/2} e^{+12.6K/T} \approx 6.55 \cdot 10^6 T^{-1.57}$ .  $\mu_n = b\mu_p = 2.07 \cdot 10^6 T^{-3/2} e^{+12.6K/T}$  ( $T$  in K), obtained from experimental data of Hall effect and resistivity. 2P, 2N: sample numbers.



**Fig. 13.**

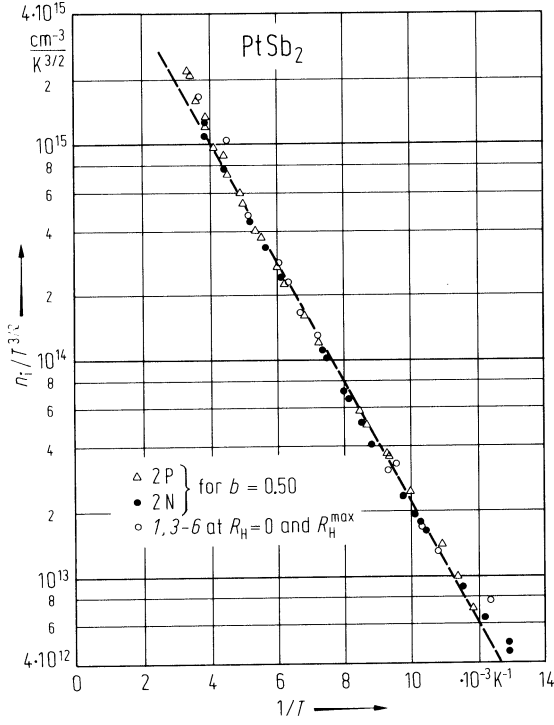
PtSb<sub>2</sub>. Hall mobility vs. temperature in a doubly-logarithmic scale [76A]. Sample numbers as in Fig. 4. At low temperatures the slopes are considerably steeper than for compensated semiconductors when electrons are scattered by ionized impurities i.e.  $R\sigma$  ceases to represent the mobility at  $T < 20$  K [76A].





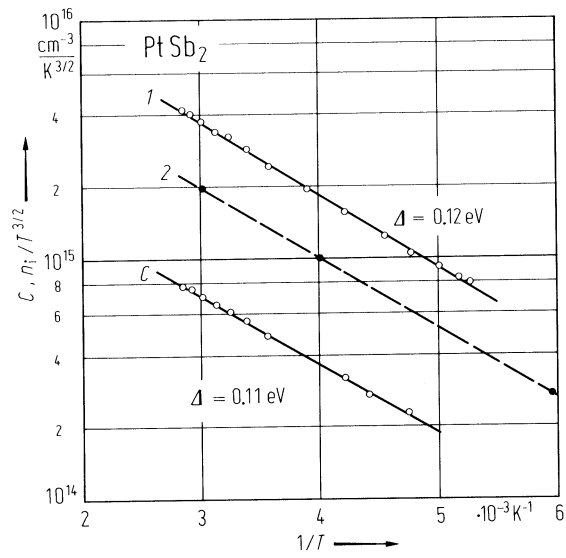
**Fig. 14.**

PtSb<sub>2</sub>.  $n_i T^{-3/2}$  vs. reciprocal temperature [68R]. For samples 2P and 2N the intrinsic carrier concentration was calculated from  $\sigma(T)$ ,  $R_H(T)$  using the relations  $n_i^2 = np$  and  $n - p = N_{\text{extr.}} = (eR_H)^{-1}$  at 77 K, and assuming  $b = \mu_n/\mu_p = 0.50$ . For samples 1, 3...6  $n_i$  was calculated at  $T$  where  $R_H = 0$ , using  $nb^2 = p = n - N_{\text{extr.}}$  as well as at the temperature at which  $R_H$  reaches a maximum, using  $nb = p = n - N_{\text{extr.}}$ . The mobility ratio  $b$  was calculated at the positive maximum of  $R_H$  for each n-type sample using the relation  $R_H^{\text{max}}/R_H^{\text{extr.}} = (b-1)^2/4b$ . The broken line represents  $n_i T^{-3/2} = 4.84 \cdot 10^{15} A e^{-E_g(0)/2kT} [\text{cm}^{-3}\text{K}^{-3/2}]$ , where  $E_g(0) = 0.110$  eV and  $A = (m_n m_p / m_0^2)^{3/4} e^{-\beta/2k} = 2.72$  ( $m_{n,p}$ : density of states masses),  $\beta$ : temperature coefficient of the band gap ( $E_g = E_g(0) + \beta T$ ).



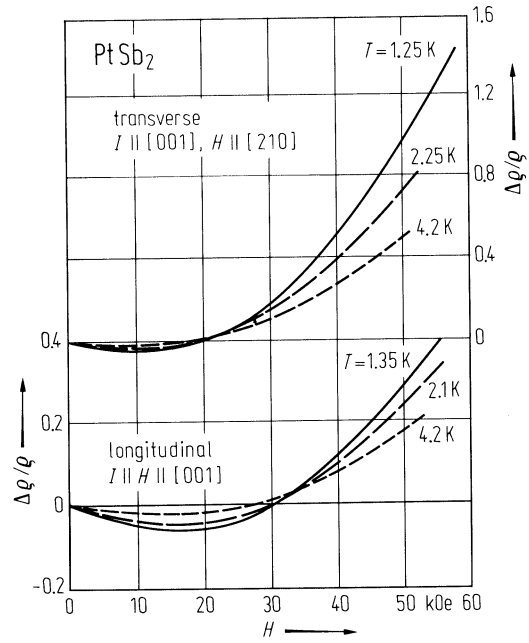
**Fig. 15.**

PtSb<sub>2</sub>. Intrinsic carrier concentration vs. reciprocal temperature [73A]. 1:  $n_i T^{-3/2}$  of [73A] calculated from Hall effect and Nernst-Ettingsbausen effect, 2:  $n_i T^{-3/2}$  data taken from [68R].  $C = n_i T^{-3/2} (m_n m_p / m_0^2)^{-3/4}$ .



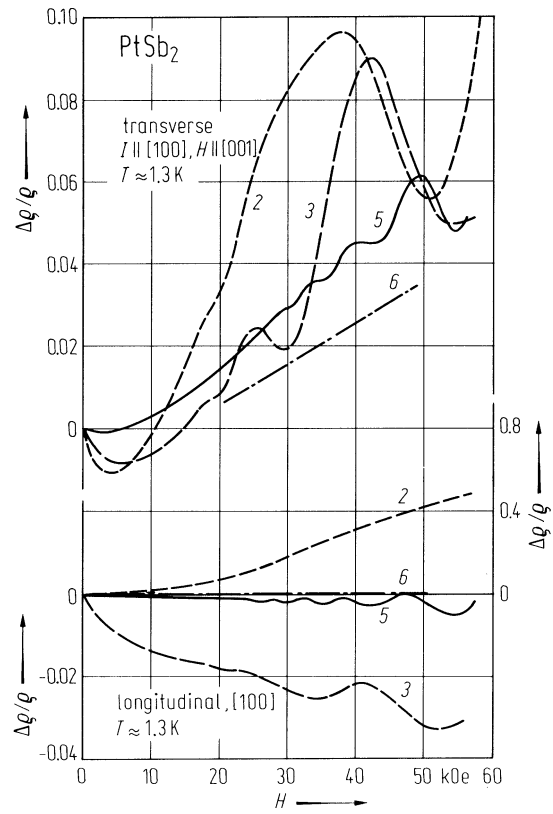
**Fig. 16.**

PtSb<sub>2</sub>. Transverse and longitudinal magnetoresistance vs. magnetic field strength [72D]. Monocrystalline p-type sample of low carrier concentration;  $R_H = 150 \text{ cm}^3/\text{As}$ ,  $\rho = 0.169 \text{ } \Omega\text{cm}$ ,  $\mu_H = 890 \text{ cm}^2/\text{Vs}$  at 4.2 K.



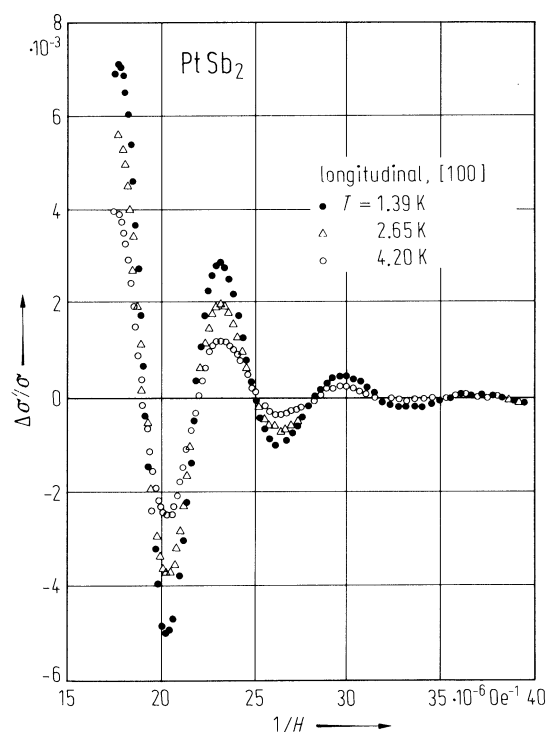
**Fig. 17.**

PtSb<sub>2</sub>. Transverse and longitudinal magnetoresistance vs. magnetic field strength [72D]. Mono-crystalline p-type samples with moderate to high carrier concentrations. 2: Sn-doped, 3,5,6: Rh-doped.



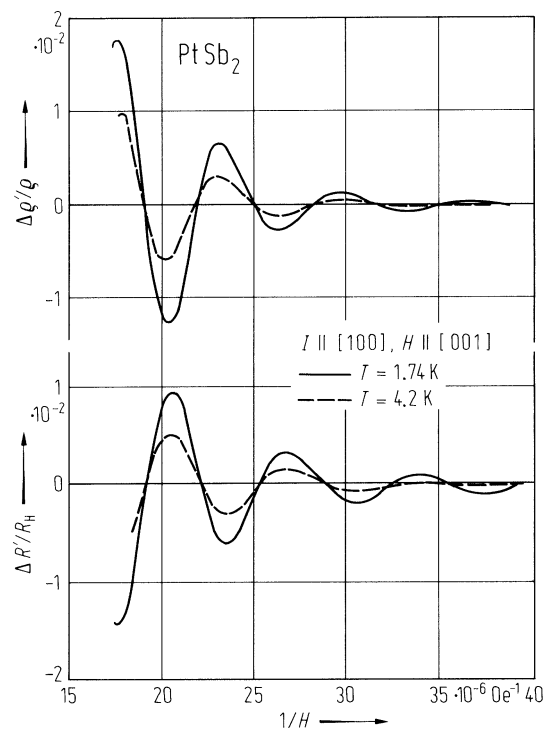
**Fig. 18.**

PtSb<sub>2</sub>. Oscillatory part of the longitudinal magnetoconductivity  $\Delta\sigma/\sigma$  vs. reciprocal magnetic field [72D]. Longitudinal effect parallel to [100]. Rh-doped p-type single crystal with  $R_H(4.2\text{ K}) = 1.87\text{ cm}^3/\text{As}$ ,  $\rho(4.2\text{ K}) = 9.14 \cdot 10^{-4}\ \Omega\text{cm}$ .



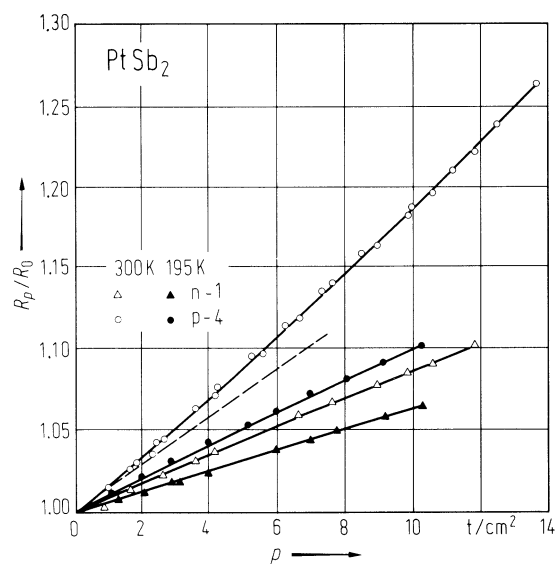
**Fig. 19.**

PtSb<sub>2</sub>. Oscillatory part of the transverse magnetoresistance  $\Delta\rho'/\rho$  and Hall coefficient  $\Delta R'/R$  vs. reciprocal magnetic field strength [72D]. For sample data, see Fig.18.



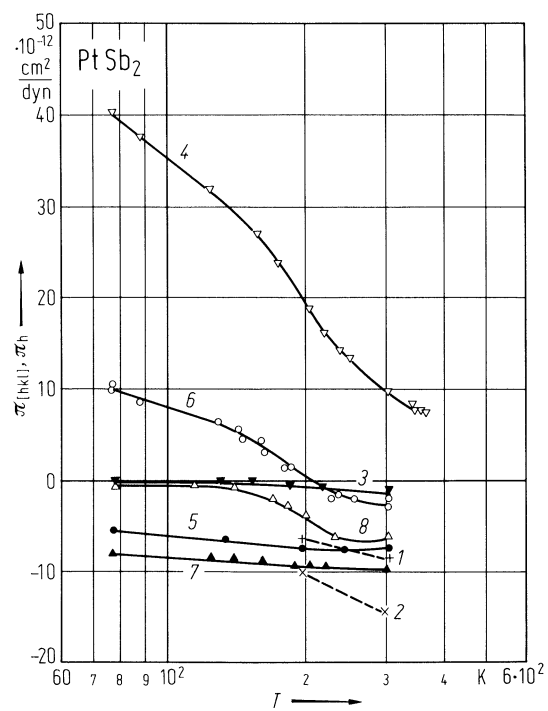
**Fig. 20.**

PtSb<sub>2</sub>. Resistance ratio  $R_p/R_0$  vs. hydrostatic pressure [65D]. ( $1 \text{ t/cm}^2 = 9.807 \cdot 10^7 \text{ Pa}$ ). Triangles and circles: n-type crystal n-1 and p-type crystal p-4, respectively, measured at 195 K (full symbols) and 300 K (open symbols).



**Fig. 21.**

PtSb<sub>2</sub>. Piezoresistance coefficients vs. temperature for various samples (see Fig. 6) [65D].





**Fig. 22**

PtSb<sub>2</sub>. Thermal conductivity vs. temperature (a) and electrical resistivity vs. temperature (b), both in doubly-logarithmic scale, measured on single crystals of unknown orientation. The resistivity curves serve to characterize the different samples [82K].

