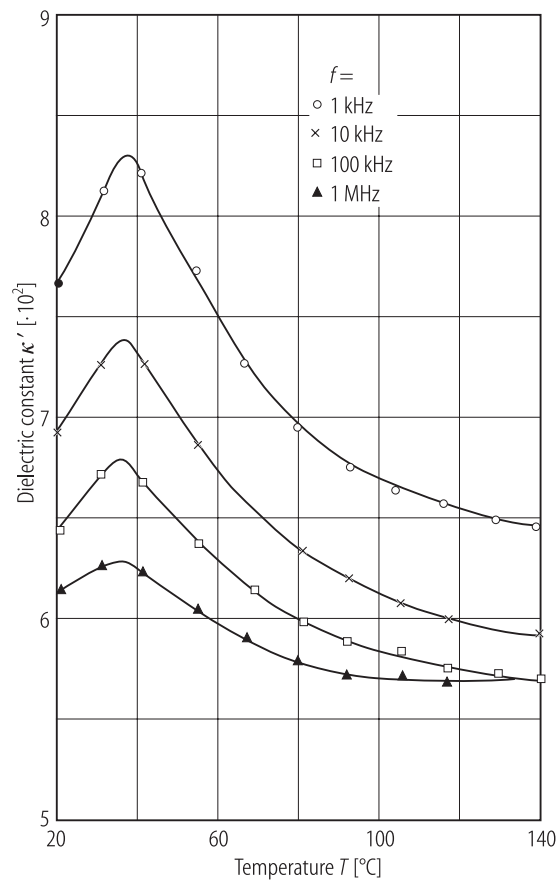
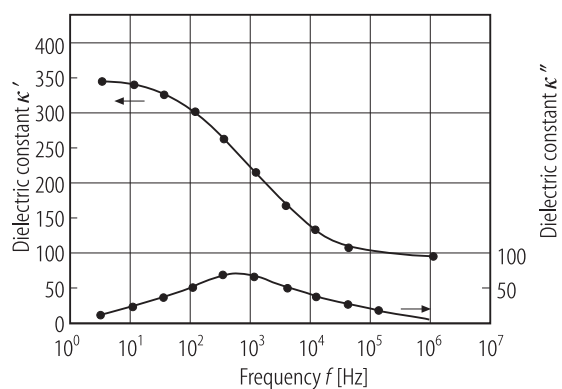


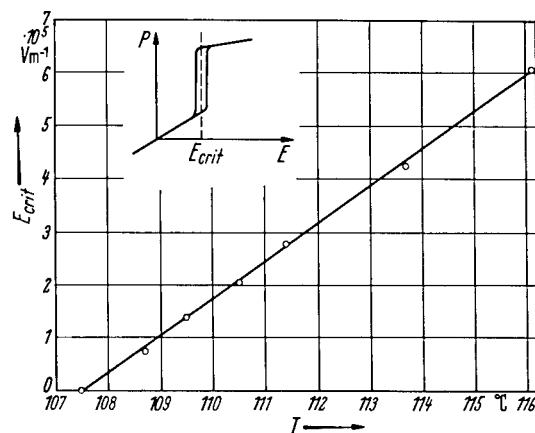
**Fig. 1A-10-060.**  $\text{BaTiO}_3$  (thin film).  $\kappa$  vs.  $T$  [93Des].  
 $f = 1$  MHz. Film thickness: 220 nm. Substrate: Si single crystal. Deposition temperature: 650 °C. Deposition pressure: 2 Pa.



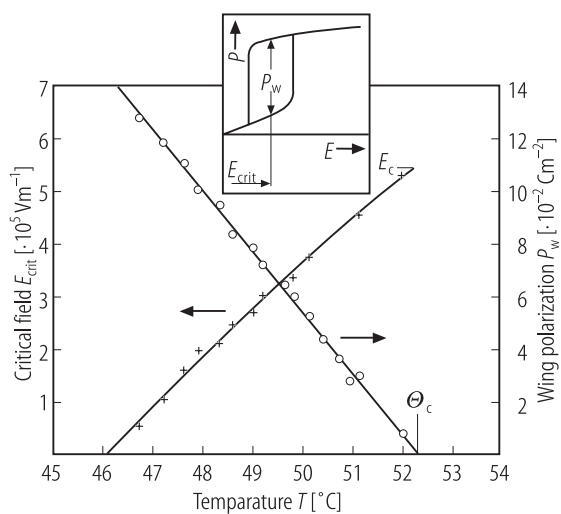
**Fig. 1A-10-061.** BaTiO<sub>3</sub> (thin film).  $\kappa'$  vs.  $T$  [93Kam].  
 Parameter:  $f$ . The thin film was grown by sol-gel method on the platinum substrate. Film thickness: 2.5  $\mu\text{m}$ .



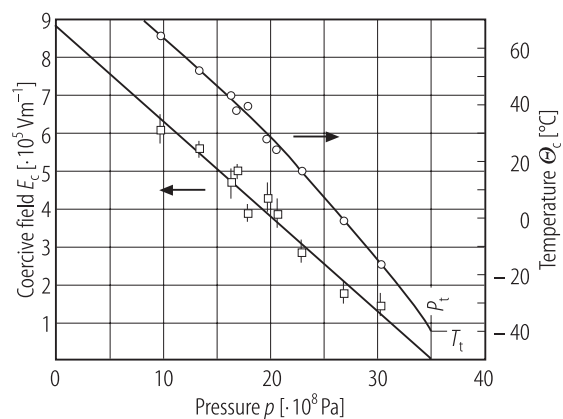
**Fig. 1A-10-062.**  $\text{BaTiO}_3$  (thin film).  $\kappa'$ ,  $\kappa''$  vs.  $f$  [93Kam]. The thin film was grown by sol-gel method on stainless-steel plate. Film thickness: 1  $\mu\text{m}$ .



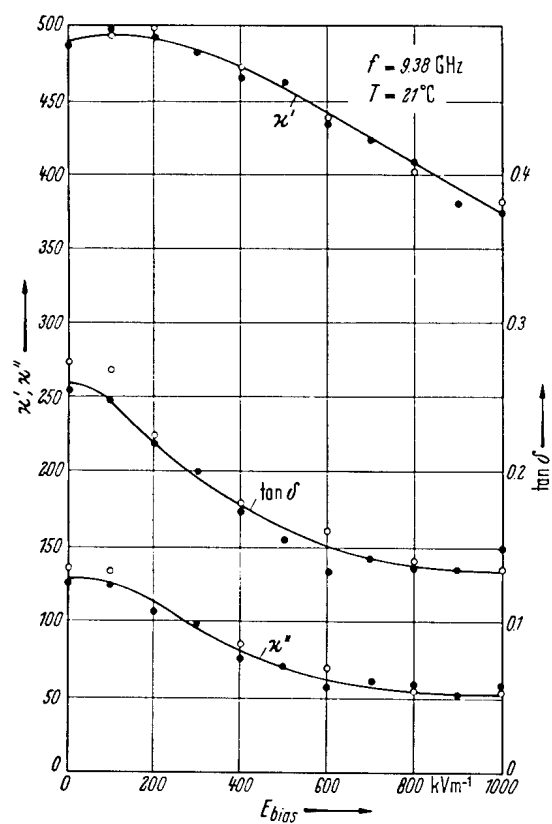
**Fig. 1A-10-063.** BaTiO<sub>3</sub>.  $E_{crit}$  vs.  $T$  [53Mer].  $E_{crit}$ : electric field strength associated with the induced P-F phase transition above the Curie temperature. For definition, see the insert.



**Fig. 1A-10-064.** BaTiO<sub>3</sub>.  $E_{\text{crit}}$ ,  $P_w$  vs.  $T$  at  $13.3 \cdot 10^8$  Pa [89Dec].  $E_{\text{crit}}$ : electric field strength associated with the induced P–F phase transition.  $P_w$ : wing polarization. For definition, see the insert.  $(E_c, \Theta_c)$ : critical point in the  $E$ – $T$  plane.



**Fig. 1A-10-065.** BaTiO<sub>3</sub>.  $E_c$ ,  $\Theta_c$  vs.  $p$  [89Dec]. ( $E_c$ ,  $\Theta_c$ ): critical point in the  $E$ - $T$  plane. ( $p_t$ ,  $T_t$ ): ( $p$ ,  $T$ ) of tricritical point.



**Fig. 1A-10-066.** BaTiO<sub>3</sub> (ceramics).  $\kappa'$ ,  $\kappa''$ ,  $\tan \delta$  vs.  $E_{\text{bias}}$   
[62Pop].  $f = 9.38 \text{ GHz}$ .  $T = 21^\circ\text{C}$ .

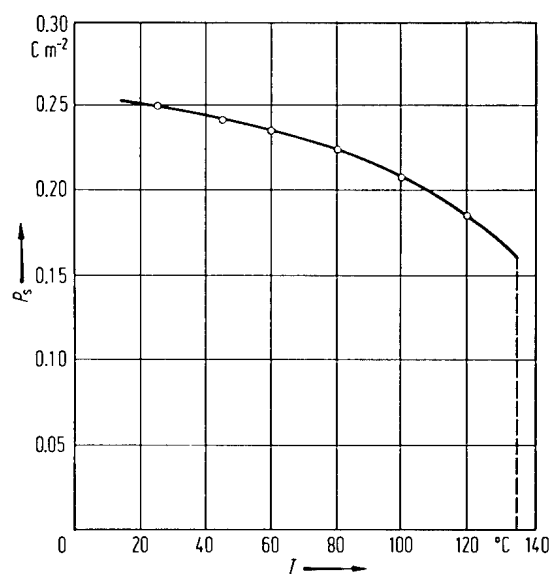
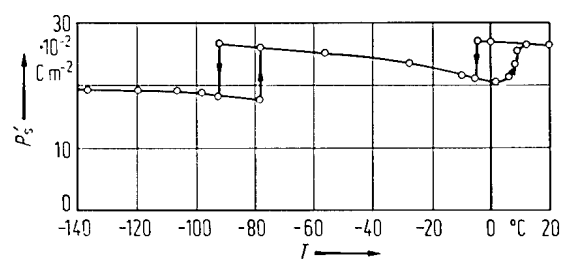
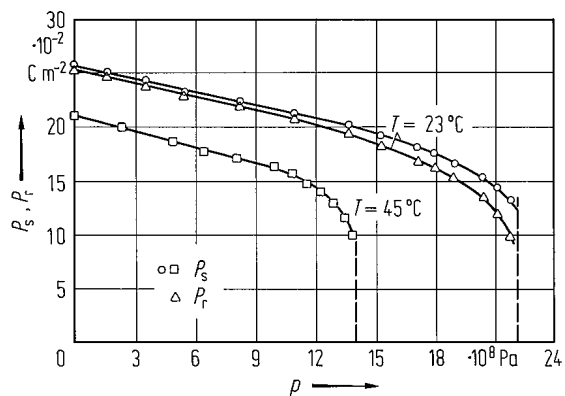


Fig. 1A-10-067. BaTiO<sub>3</sub>.  $P_s$  vs.  $T$  [68Wem].

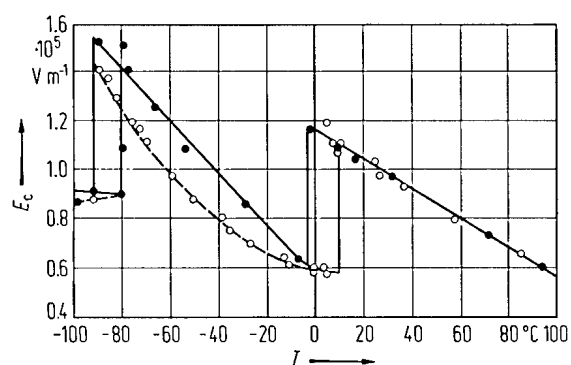




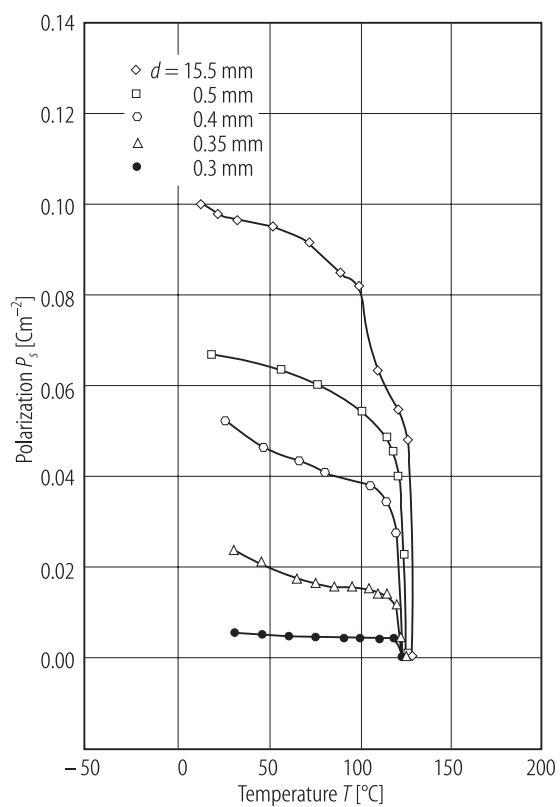
**Fig. 1A-10-068.** BaTiO<sub>3</sub>.  $P'_s$  vs.  $T$  below RT [55Wie1].  $P'_s$ :  $P'_s = P_s$  in the phase II, and component of  $P_s$  along the pseudotetragonal [001] in the phase III.



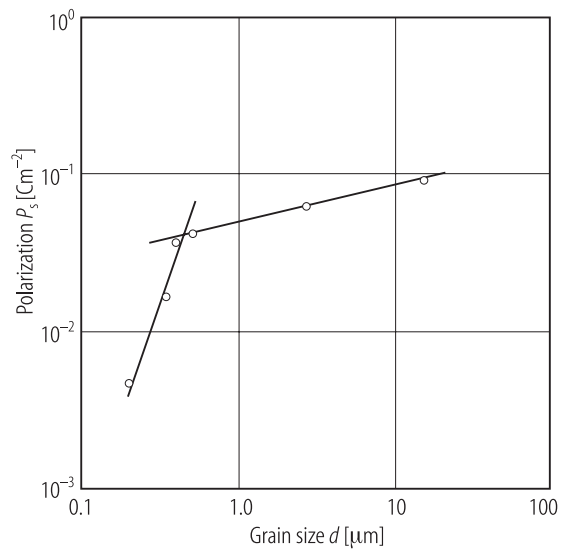
**Fig. 1A-10-069.** BaTiO<sub>3</sub>.  $P_s, P_r$  vs.  $p$  at  $23^\circ\text{C}$  and  $45^\circ\text{C}$  [66Sam].  $P_r$ : remanent polarization.



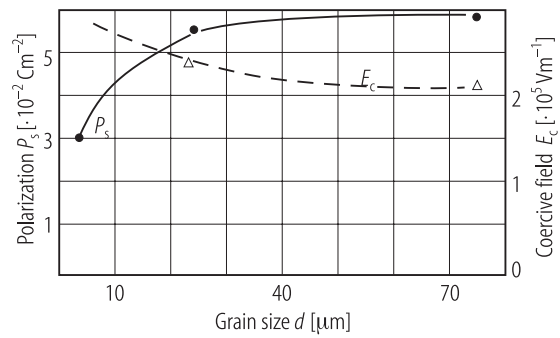
**Fig. 1A-10-070.**  $\text{BaTiO}_3$ .  $E_c$  vs.  $T$  [55Wie2]. Solid curve:  $c$ -domain crystal at 25 °C. Dashed curve:  $c$ -domain crystal having a number of  $a$ -domain wedges at 25 °C.  $E_c$  was measured with plates cut perpendicularly to the pseudocubic [001].



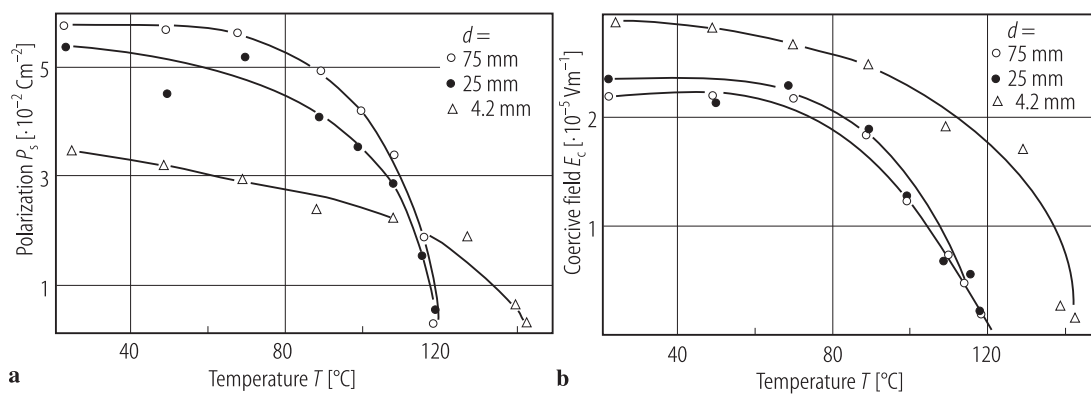
**Fig. 1A-10-071.** BaTiO<sub>3</sub> (ceramics).  $P_s$  vs.  $T$  [86Sha].  
Parameter:  $d$ , average grain size.



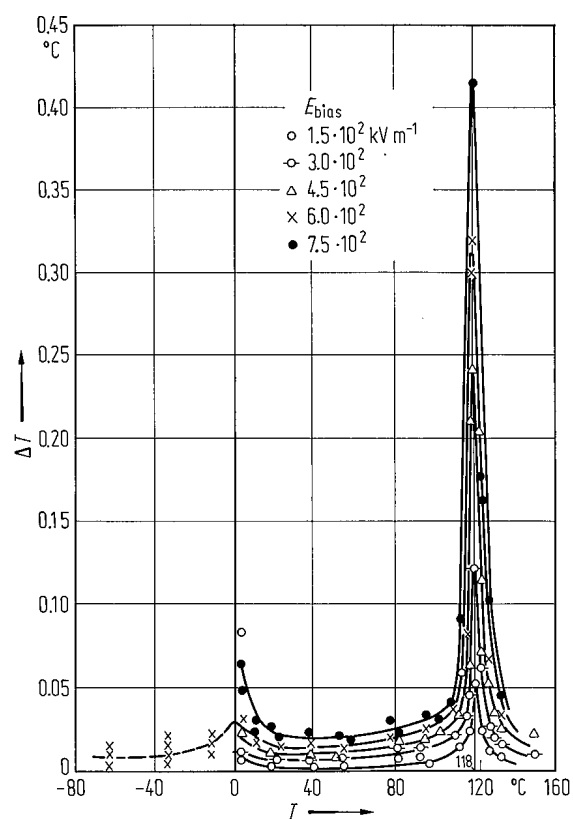
**Fig. 1A-10-072.**  $\text{BaTiO}_3$  (ceramics).  $P_s$  vs.  $d$  at  $70^\circ\text{C}$  [89Sha].  $d$ : average grain size.



**Fig. 1A-10-073.** BaTiO<sub>3</sub> (ceramics).  $P_s$ ,  $E_c$  vs.  $d$  [81Dam].  
 $d$ : thickness of the specimens.  $T = \text{RT}$ .

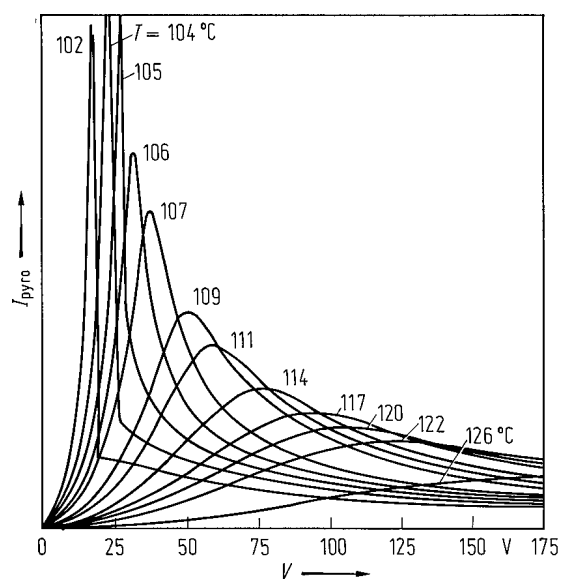


**Fig. 1A-10-074.** BaTiO<sub>3</sub> (ceramics).  $P_s$  (a),  $E_c$  (b) vs.  $T$  [81Dam]. Parameter:  $d$ , thickness of the specimens.

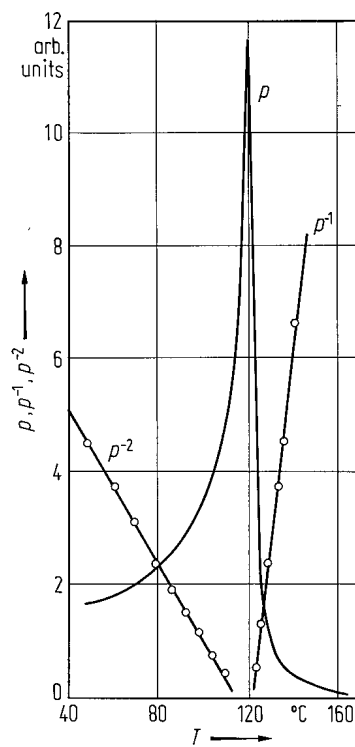


**Fig. 1A-10-075.** BaTiO<sub>3</sub> (ceramics).  $\Delta T$  vs.  $T$  [61Kar].  
Parameter:  $E_{\text{bias}}$ .  $\Delta T$ : electrocaloric temperature change.

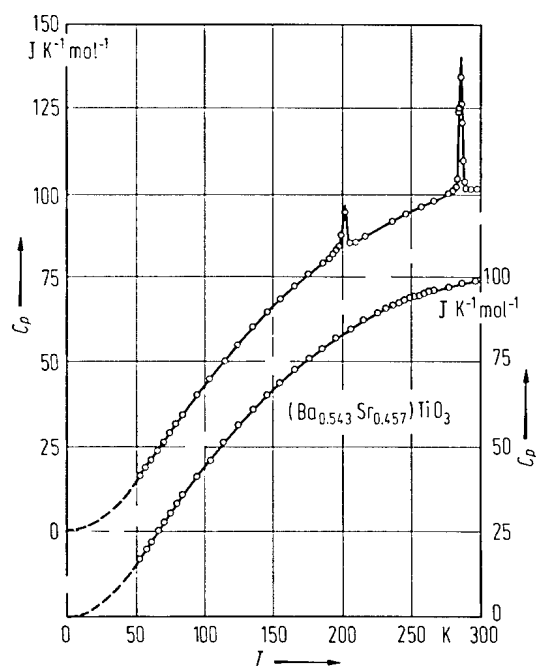




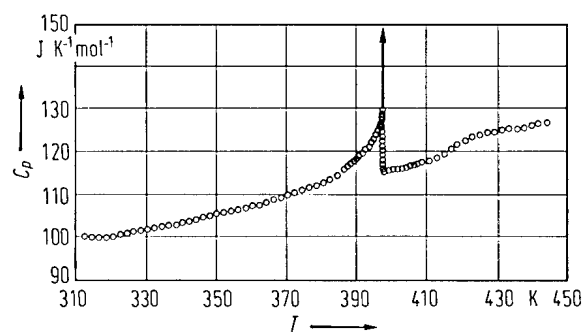
**Fig. 1A-10-076.**  $\text{BaTiO}_3$ .  $I_{\text{pyro}}$  vs.  $V$  [56Chy1]. Parameter:  $T$ .  $I_{\text{pyro}}$ : pyroelectric current measured by the dynamic method.



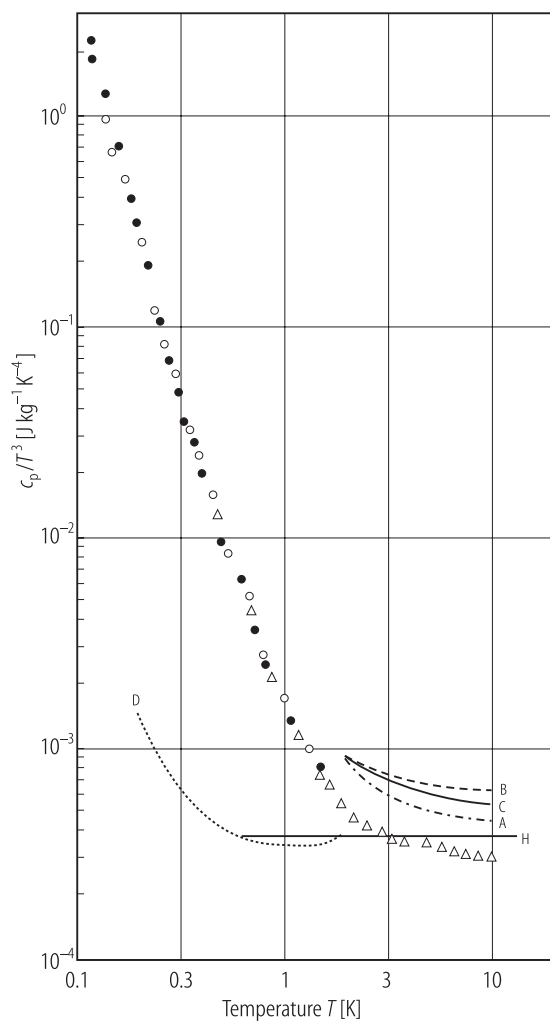
**Fig. 1A-10-077.** BaTiO<sub>3</sub> (ceramics).  $p$ ,  $p^{-1}$ ,  $p^{-2}$  vs.  $T$  [77Toy].  $p$ : pyroelectric coefficient.



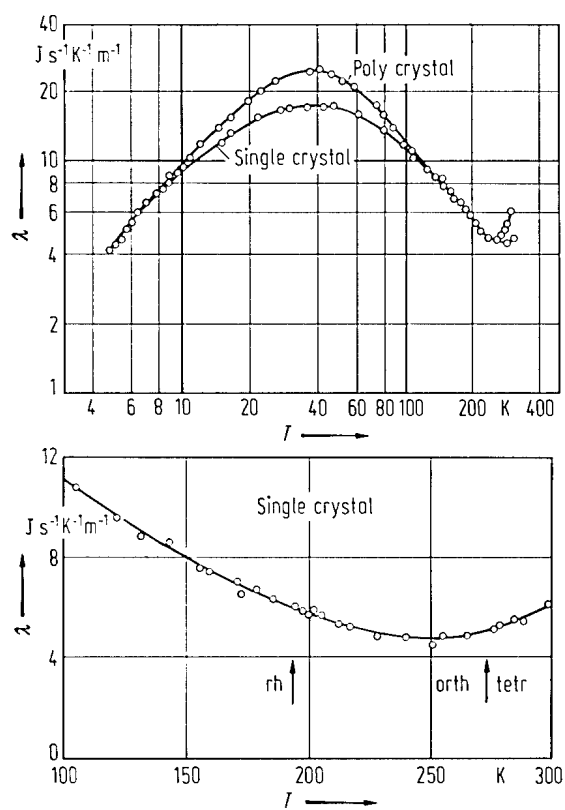
**Fig. 1A-10-078.**  $\text{BaTiO}_3$  (ceramics).  $\text{Ba}_{1-x}\text{Sr}_x\text{TiO}_3$ .  $C_p$  vs.  $T$  [52Tod].  $C_p$ : molar heat capacity at constant pressure.



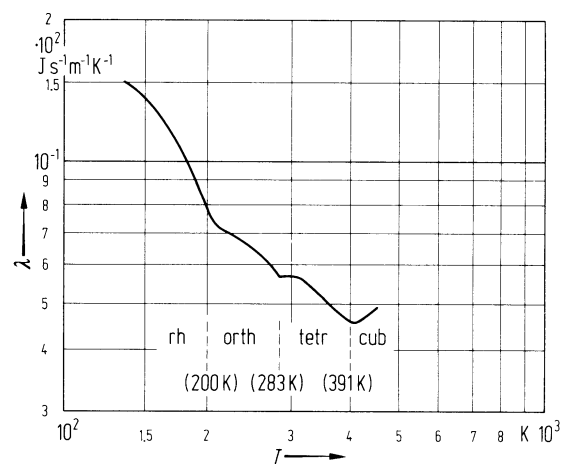
**Fig. 1A-10-079.** BaTiO<sub>3</sub>.  $C_p$  vs.  $T$  [76Hat].  $C_p$ : molar heat capacity at constant pressure.



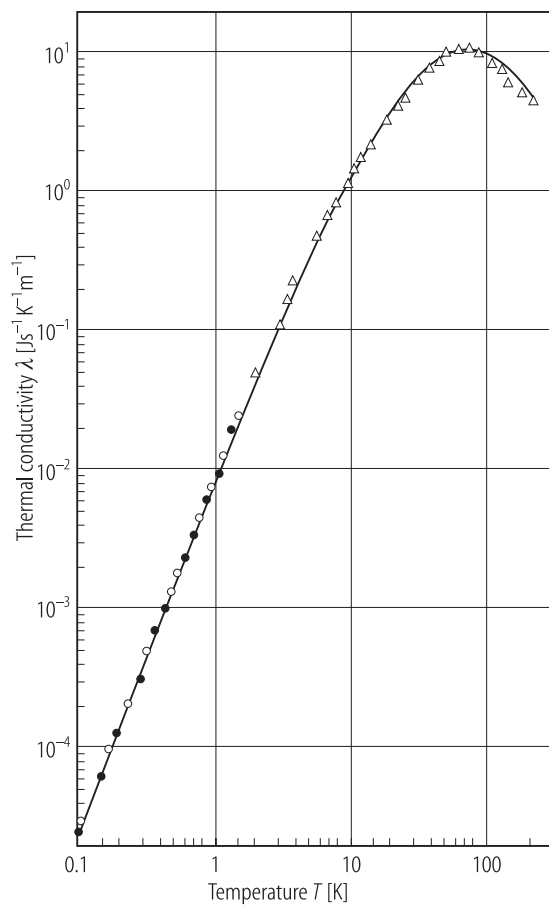
**Fig. 1A-10-080.** BaTiO<sub>3</sub> (ceramics).  $c_p/T^3$  vs.  $T$  below 10 K [86Zim].  $c_p$ : specific heat capacity at constant pressure. Circles, open triangles : present data of ceramics. A: poled ceramics [77Law]. B: unpoled ceramics [77Law]. C: single crystal [77Law]. D: single crystal [85Foo]. H: calculated results of Debye-contribution [78Law].



**Fig. 1A-10-081.** BaTiO<sub>3</sub>.  $\lambda$  vs.  $T$  [65Sue].  $\lambda$ : thermal conductivity.

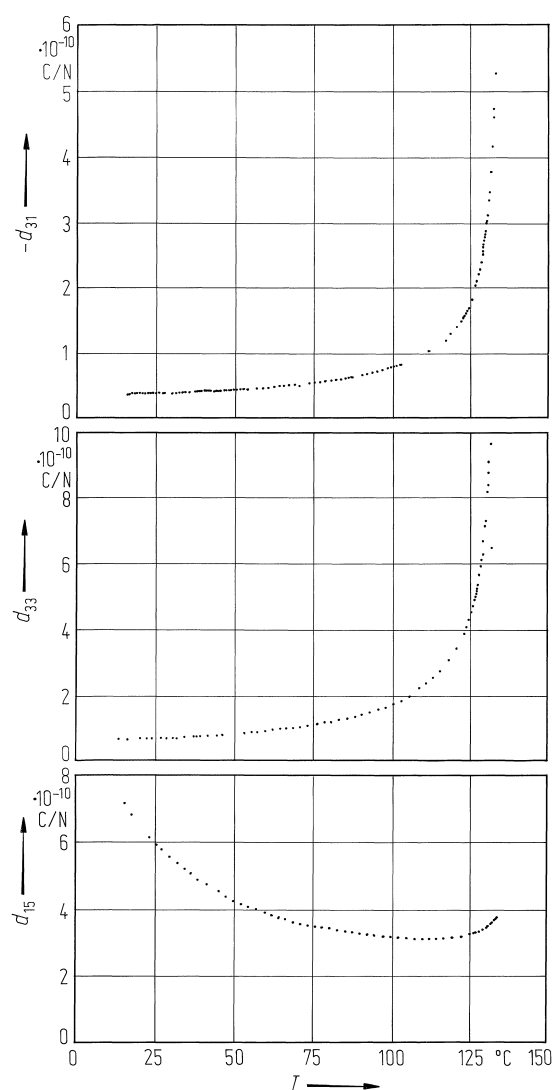


**Fig. 1A-10-082.**  $\text{BaTiO}_3$ .  $\lambda$  vs.  $T$  [67Man].  $\lambda$ : thermal conductivity.

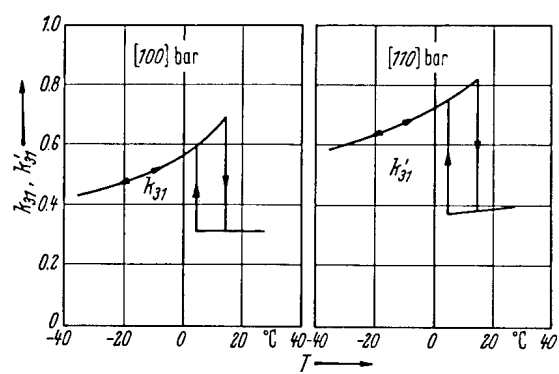


**Fig. 1A-10-083.** BaTiO<sub>3</sub> (ceramics).  $\lambda$  vs.  $T$  [86Zim].  $\lambda$ : thermal conductivity. Open circles: unpolarized. Full circles: polarized. Open triangles: without any treatments.

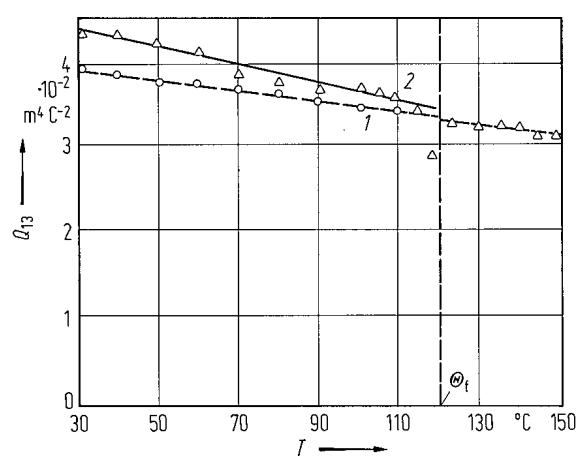




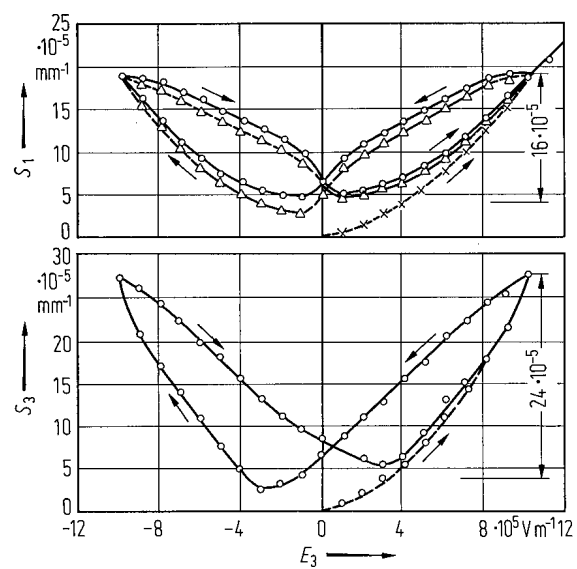
**Fig. 1A-10-084.**  $\text{BaTiO}_3$ .  $d_{i\lambda}$  vs.  $T$  [86Sch]. The crystal was grown by top-seeded solution method.



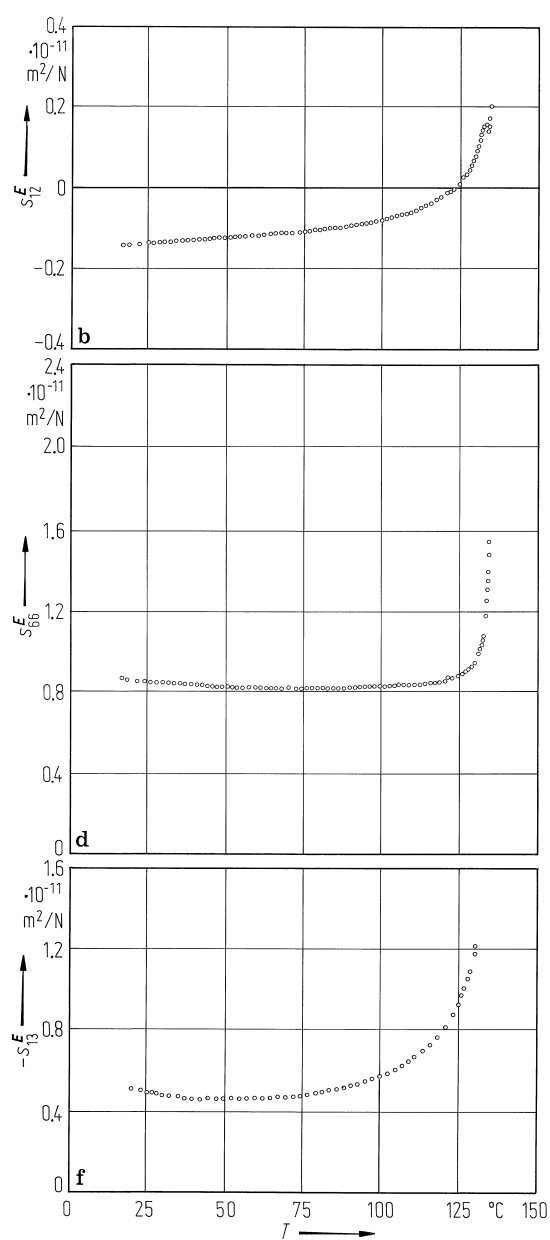
**Fig. 1A-10-085.** BaTiO<sub>3</sub>.  $k_{31}$ ,  $k'_{31}$  for [100] and [110] bars vs.  $T$  [58Ber].



**Fig. 1A-10-086.**  $\text{BaTiO}_3$ .  $Q_{13}$  vs.  $T$  [59Hui].  $Q_{13}$ : electrostrictive constant. Dashed line 1: calculated from  $S_1/P_3^2$  ( $S_1$ : determined from X-ray,  $P_3$ : determined from hysteresis loops). Solid line 2: calculated from  $g_{31}/(2P_3)$ .



**Fig. 1A-10-087.** BaTiO<sub>3</sub> (ceramics).  $S_1$ ,  $S_3$  vs.  $E_3$  [57Tak].  
 $S_1$ : transverse strain.  $S_3$ : longitudinal strain.



**Fig. 1A-10-088.**  $\text{BaTiO}_3$ .  $s_{\lambda\mu}^E$  vs.  $T$  [86Sch]. The crystal was grown by top-seeded solution method.

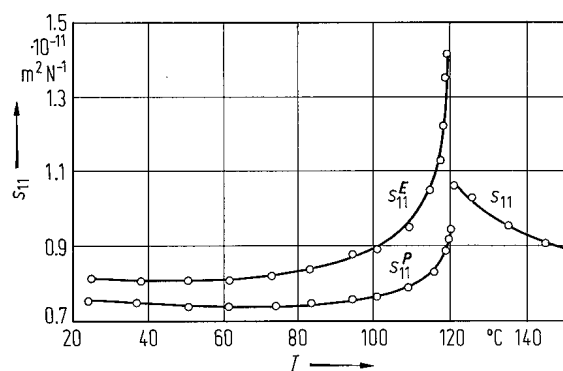
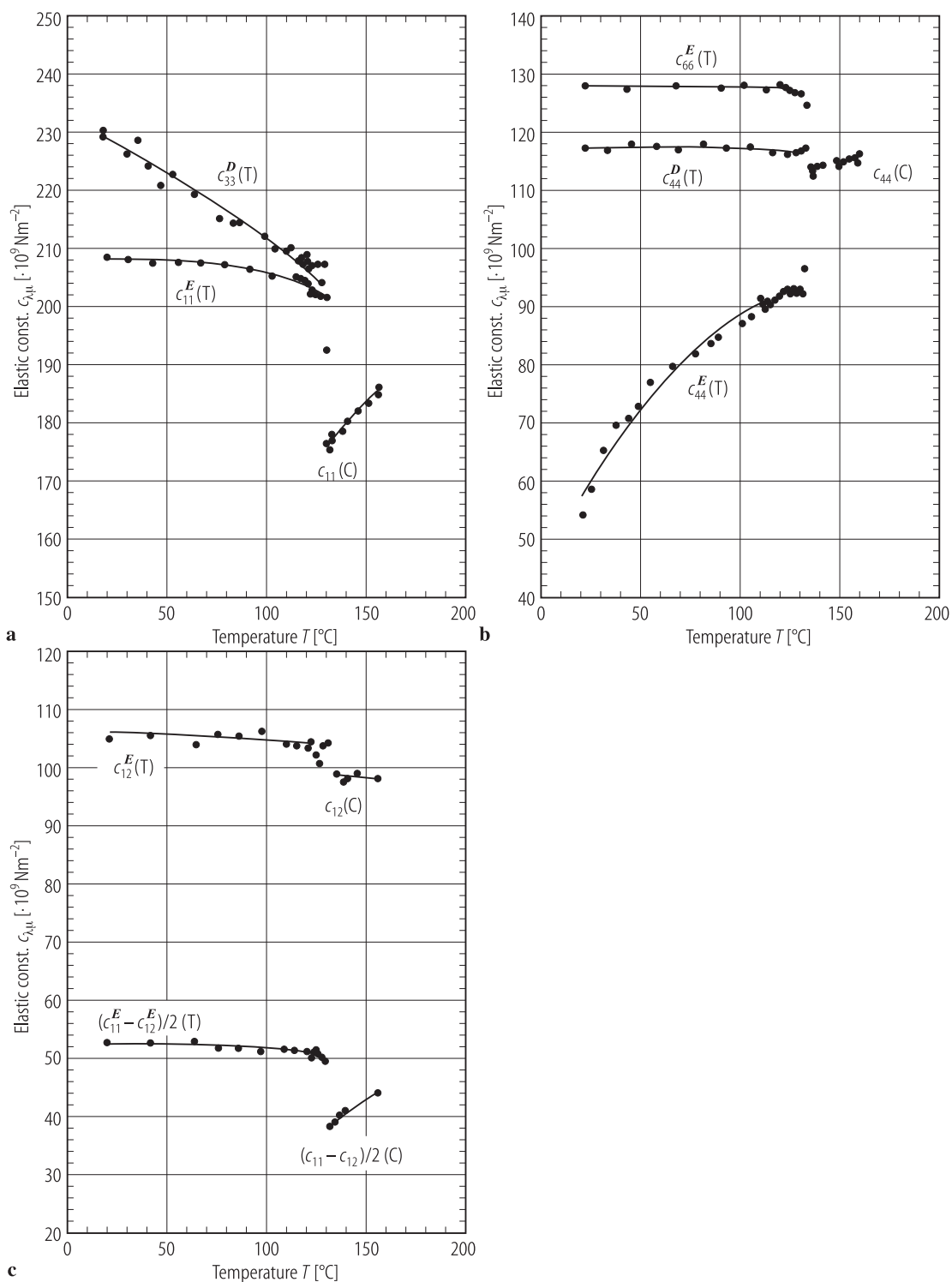
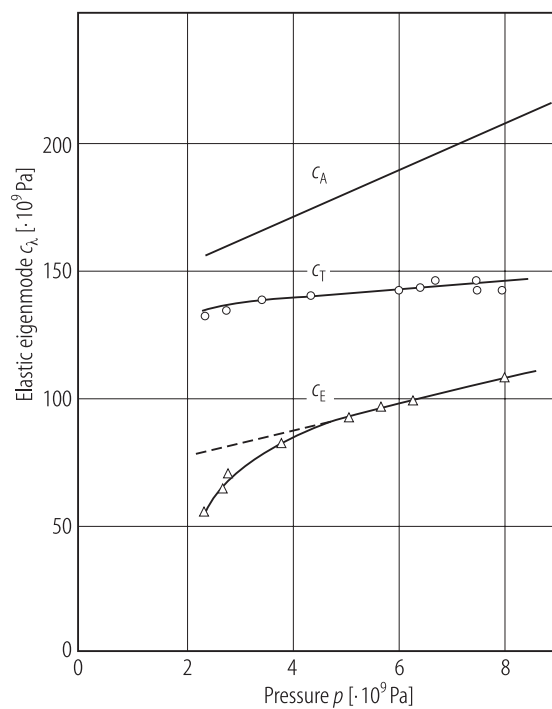


Fig. 1A-10-089. BaTiO<sub>3</sub>.  $s_{11}$  vs.  $T$  [59Hui].

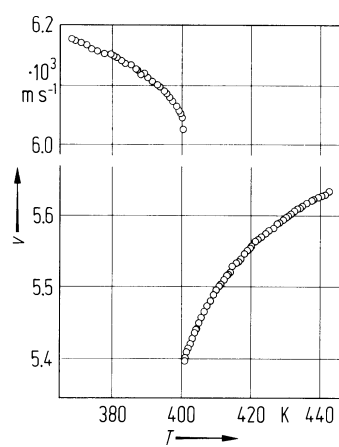


**Fig. 1A-10-090.**  $\text{BaTiO}_3$ .  $c_{\lambda\mu}$  vs.  $T$  [94LiZ]. (a)  $c_{11}$ ,  $c_{33}$ . (b)  $c_{44}$ ,  $c_{66}$ . (c)  $c_{11}$ ,  $c_{12}$ .

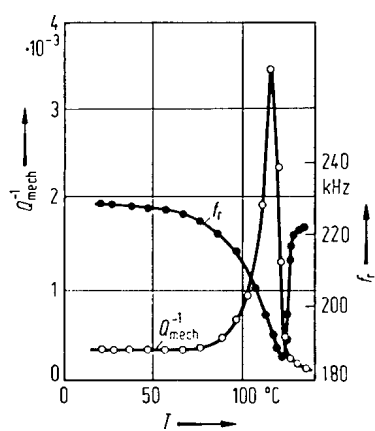


**Fig. 1A-10-091.** BaTiO<sub>3</sub>.  $c_\lambda$  vs.  $p$  [89Ish].  $c_\lambda$ : elastic eigenmode in the cubic phase.  $c_A = (c_{11} + 2c_{12})/3$ .  $c_E = (c_{11} - c_{12})/2$ .  $c_T = c_{44}$ .  $c_A$  was evaluated from observed  $c_{11}$  and  $c_E$ .

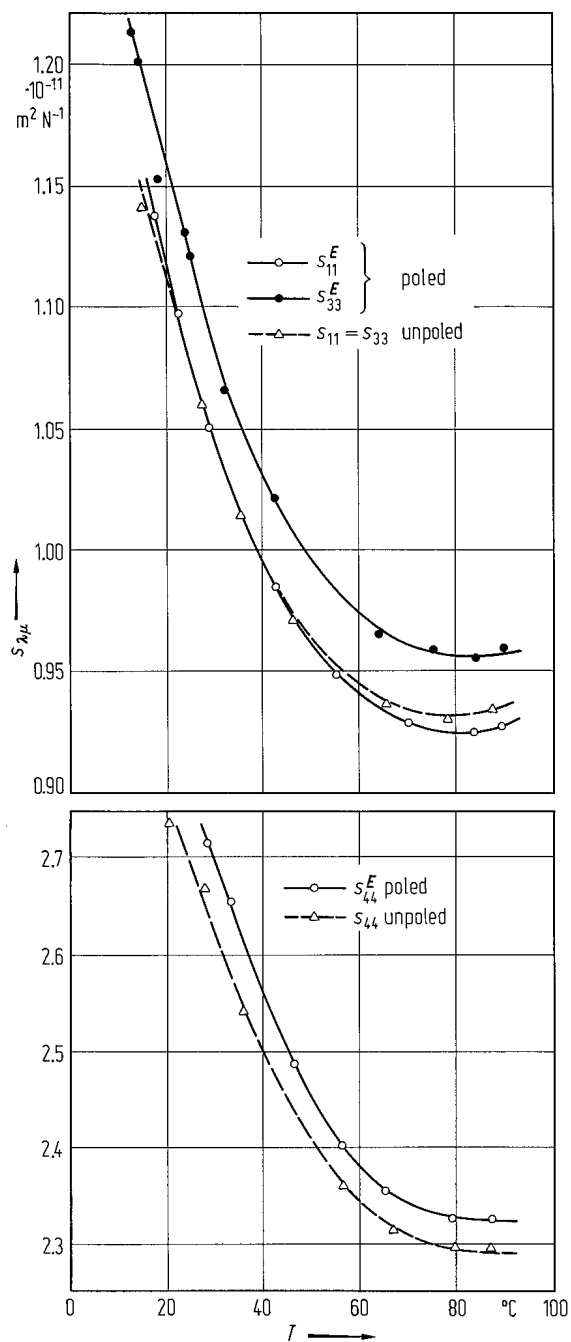




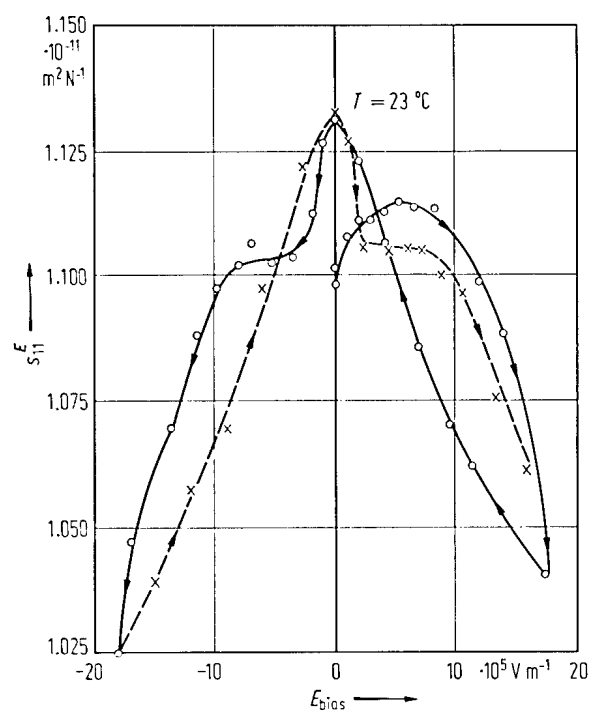
**Fig. 1A-10-092.** BaTiO<sub>3</sub>.  $v$  vs.  $T$  [73Kas].  $v$ : longitudinal sound velocity along [100] direction measured at 45 MHz.



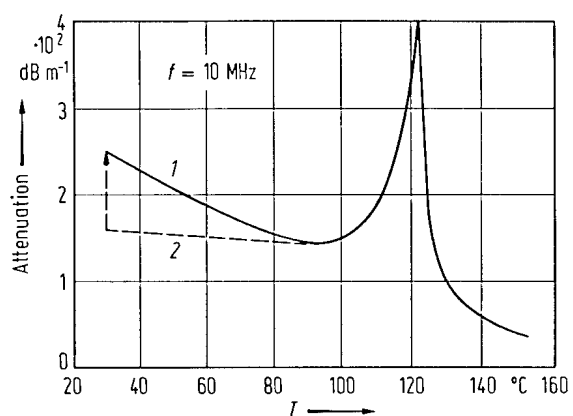
**Fig. 1A-10-093.** BaTiO<sub>3</sub>.  $f_r$ ,  $Q_{\text{mech}}^{-1}$  vs.  $T$  [69Kud].  $f_r$ : resonance frequency of the crystal along [100].



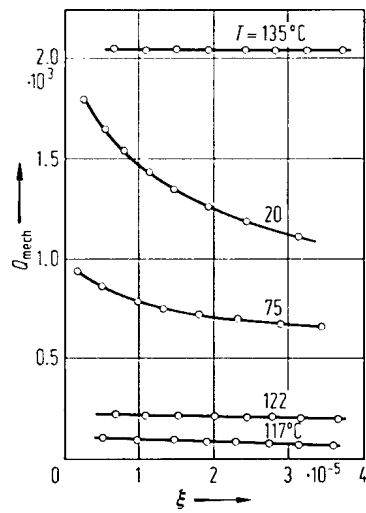
**Fig. 1A-10-094.** BaTiO<sub>3</sub> (ceramics).  $s_{11}$ ,  $s_{33}$  (poled and unpoled specimens),  $s_{44}$  (unpoled specimen),  $s_{44}^E$  (poled specimen) vs.  $T$  [57Mar].



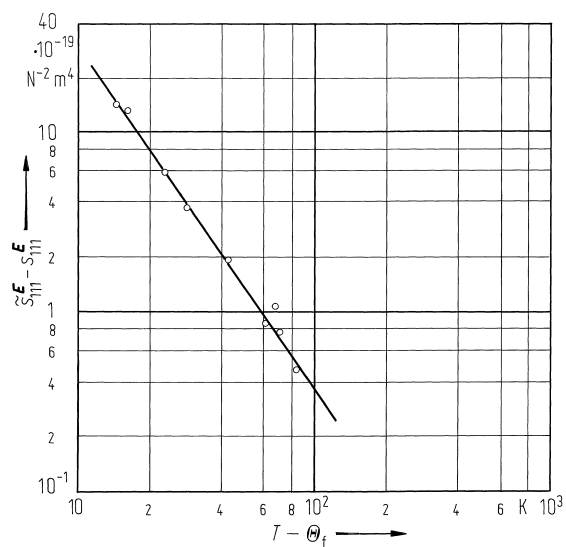
**Fig. 1A-10-095.** BaTiO<sub>3</sub> (ceramics).  $s_{11}^E$  vs.  $E_{\text{bias}}$  [57Mar].



**Fig. 1A-10-096.**  $\text{BaTiO}_3$  (ceramics). Attenuation vs.  $T$  [55Hue]. Pulse method. Curve 1: unpoled sample, heating or cooling without bias. 2: poled sample, cooling under bias.

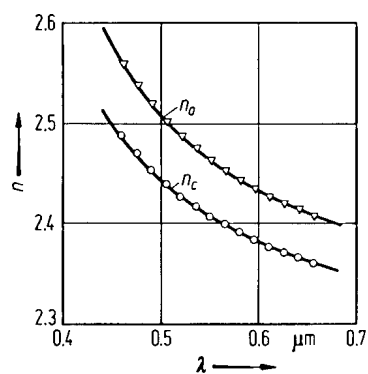


**Fig. 1A-10-097.** BaTiO<sub>3</sub>,  $Q_{\text{mech}}$  vs.  $\xi$  [71Kud]. Parameter:  $T$ .  $\xi$ : deformation amplitude.



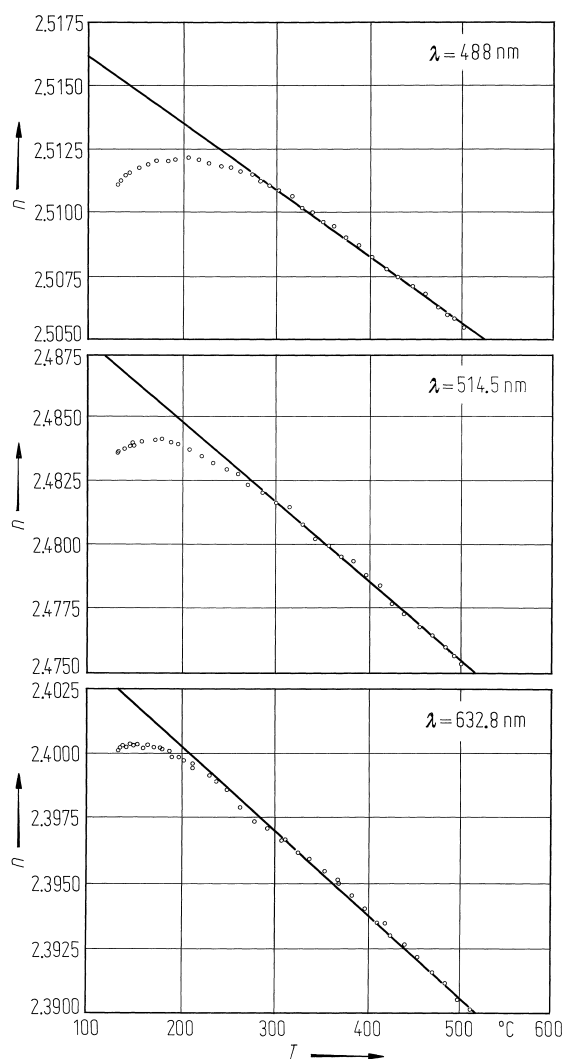
**Fig. 1A-10-098.** BaTiO<sub>3</sub>.  $\tilde{s}_{111}^E - s_{111}^E$  vs.  $T - \Theta_f$  [80Bei].

$\tilde{s}_{111}^E - s_{111}^E$ : anomalous part of the nonlinear elastic compliance.

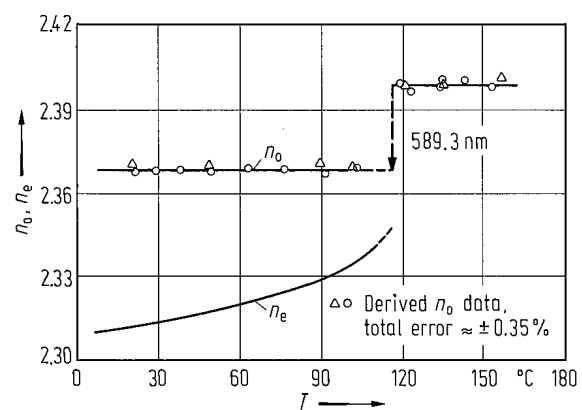


**Fig. 1A-10-099.** BaTiO<sub>3</sub>,  $n_o$  and  $n_c$  vs.  $\lambda$  at RT [71Joh]. For the crystal grown by the top-seeded solution growth method, see [68Wem]. For the crystal grown by the TiO<sub>2</sub> rich melt, see [93Bus] and Table 1A-10-018.

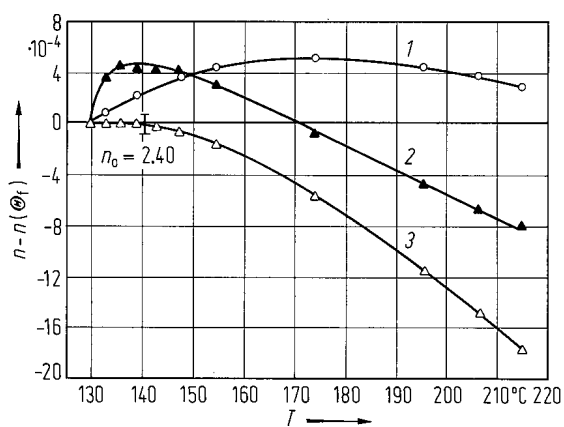




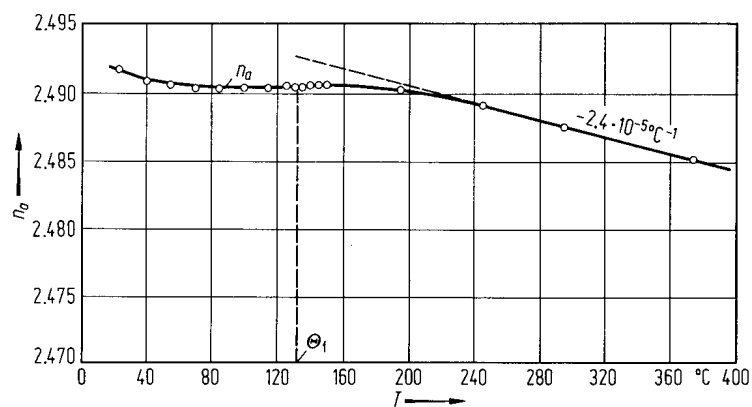
**Fig. 1A-10-100.**  $\text{BaTiO}_3$ ,  $n$  vs.  $T$  [81Bur]. Parameter:  $\lambda$ , wavelength. The crystal was grown by the  $\text{TiO}_2$  rich melt. For the flux grown crystal, see [82Bur]. For the crystals with additives of Fe, Mn, or Sr, see [90Bur].



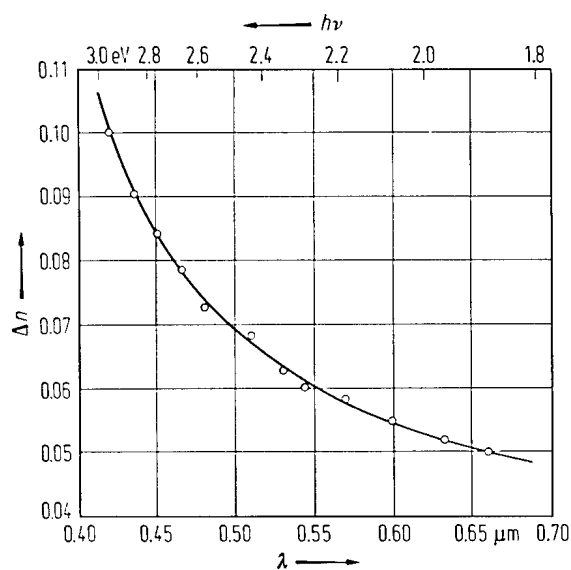
**Fig. 1A-10-101.** BaTiO<sub>3</sub>.  $n$  vs.  $T$  [64Law].  $\lambda = 589.3$  nm.



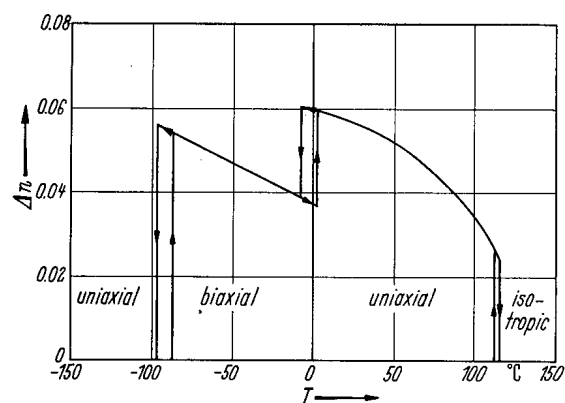
**Fig. 1A-10-102.**  $\text{BaTiO}_3$ .  $n - n(\Theta_f)$  vs.  $T$  [65Bal].  $n(\Theta_f)$ :  $n$  at  $\Theta_f$ .  $\lambda = 632.8 \text{ nm}$ . Curve 1: uncorrected, 2: corrected for thermal expansion with data of [47Meg], 3: corrected for thermal expansion with data of [52Shi].



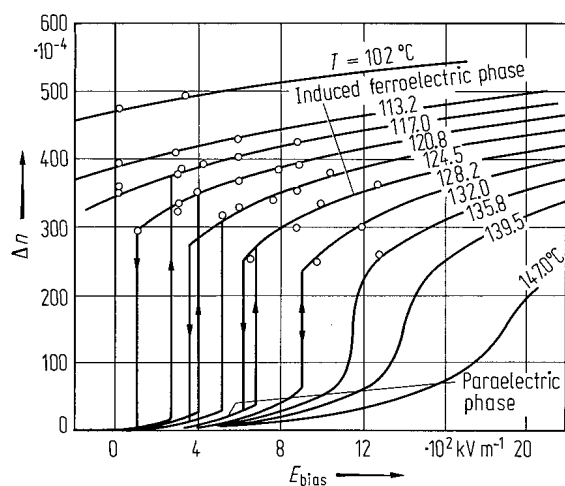
**Fig. 1A-10-103.** BaTiO<sub>3</sub>.  $n_d$  vs.  $T$  [70Wem].  $\lambda = 514.5$  nm. The crystal was grown by the top-seeded solution technique.



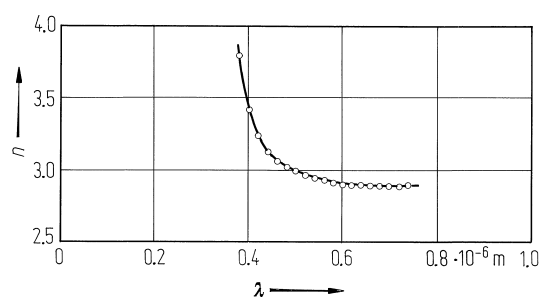
**Fig. 1A-10-104.**  $\text{BaTiO}_3$ ,  $\Delta n$  vs.  $\lambda$  [68Wem].  $T = 25^\circ\text{C}$ . The crystal was grown by the top-seeded solution technique; see also [86Rup] and [93Bus]. For the flux grown crystal, see [71Joh].



**Fig. 1A-10-105.** BaTiO<sub>3</sub>.  $\Delta n$  vs.  $T$  [49Kay].  $\Delta n = n_a - n_c$ . For green Hg line, see [58Mey]. For NaD line, see [49For]. For the crystal grown by the top-seeded solution technique, see [68Wem]. See also [88Kov].

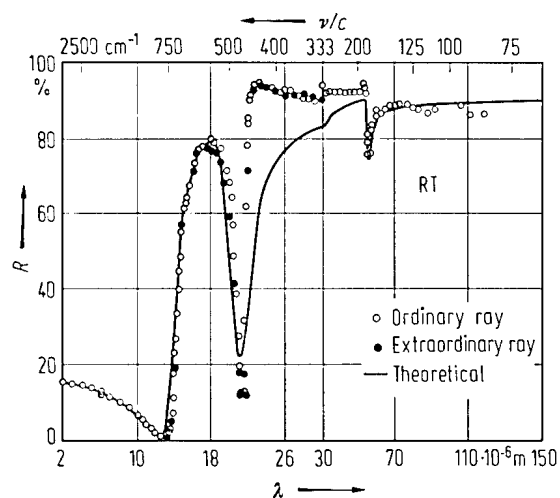


**Fig. 1A-10-106.**  $\text{BaTiO}_3$ .  $\Delta n$  vs.  $E_{\text{bias}}$  [58Mey].  
 $\Delta n = n_a - n_c$ . Parameter:  $T$ . See also [66Per].  $\lambda = 545 \text{ nm}$ .

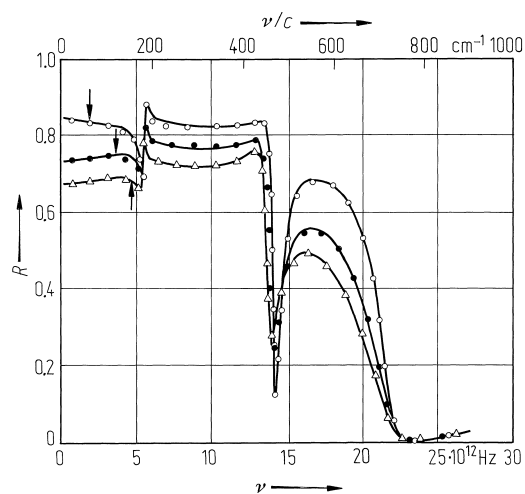


**Fig. 1A-10-107.** BaTiO<sub>3</sub> (thin film).  $n$  vs.  $\lambda$  [81Nag].

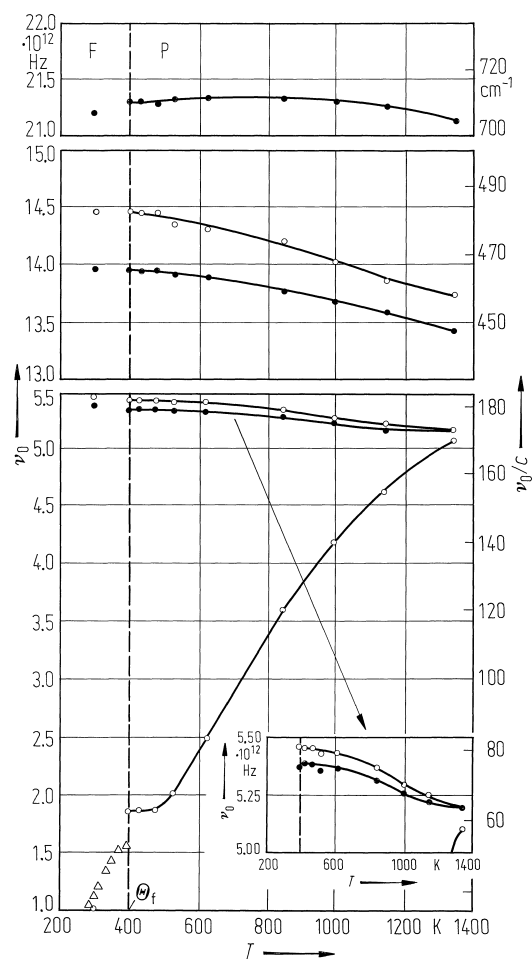




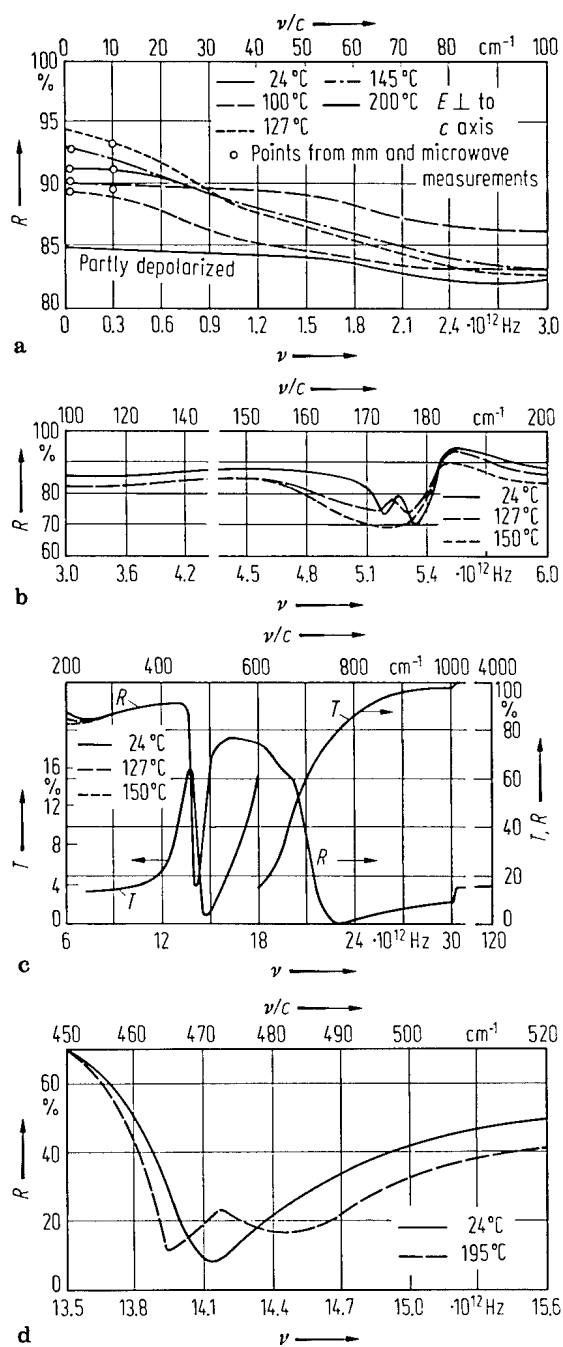
**Fig. 1A-10-108.** BaTiO<sub>3</sub>.  $R$  vs.  $\lambda$  [62Spi].  $R$ : reflectivity. See also [62Ike]. Note change of scale at  $\lambda = 30 \cdot 10^{-6}$  m. The solid curve was calculated by using dispersion theory.



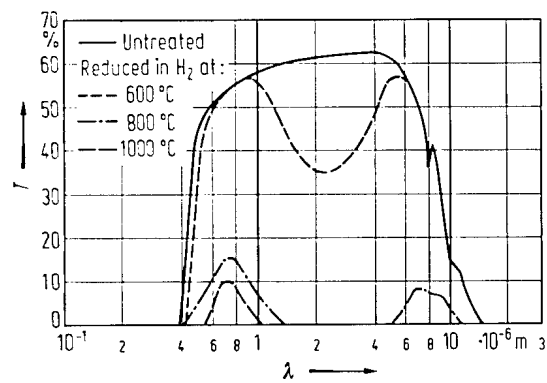
**Fig. 1A-10-109.** BaTiO<sub>3</sub>.  $R$  vs.  $\nu$  [80Lus]. Parameter:  $T$ .  $R$ : reflectivity of  $F_{1u}$ -type modes.  $\nu$ : frequency. Open circles: 400 K. Full circles: 850 K. Open triangles: 1150 K. The curves are best fit of the dielectric function model to the data. The arrows indicate the soft mode frequency.



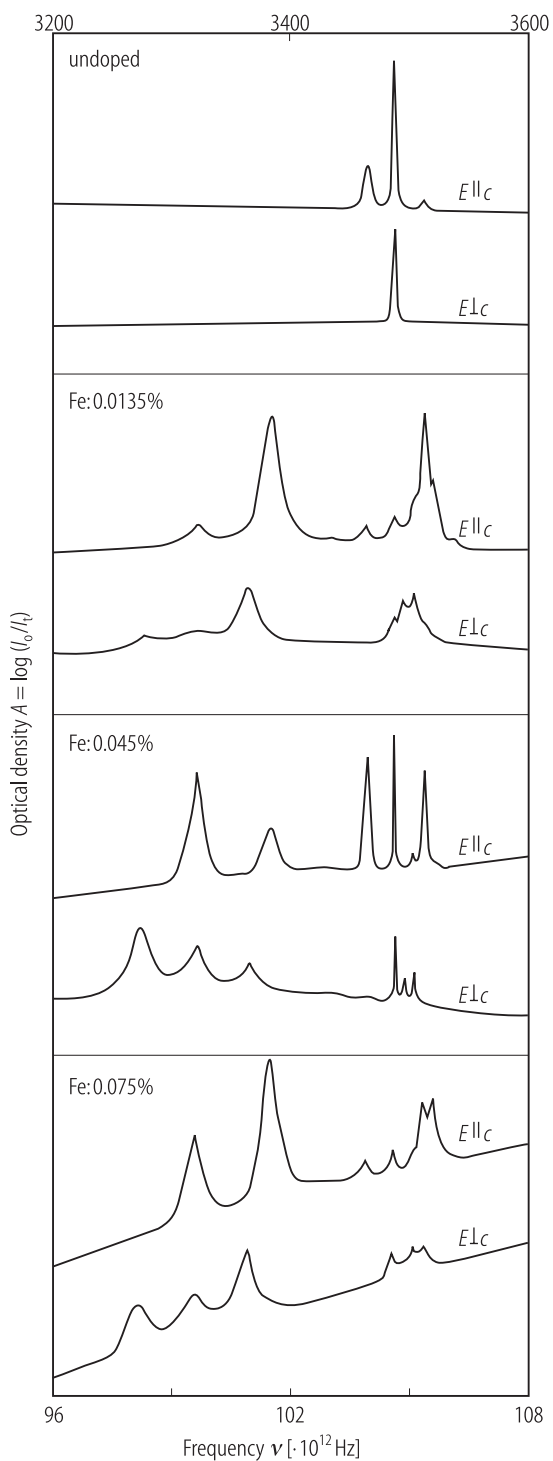
**Fig. 1A-10-110.**  $\text{BaTiO}_3$ .  $\nu_0$  vs.  $T$  [80Lus].  $\nu_0$ : frequencies of E modes below  $\Theta_f$  and  $F_{1u}$  modes above  $\Theta_f$ . Infrared measurement. Open circles: TO modes. Full circles: LO modes. Open triangles: Raman data taken from [77Sca].



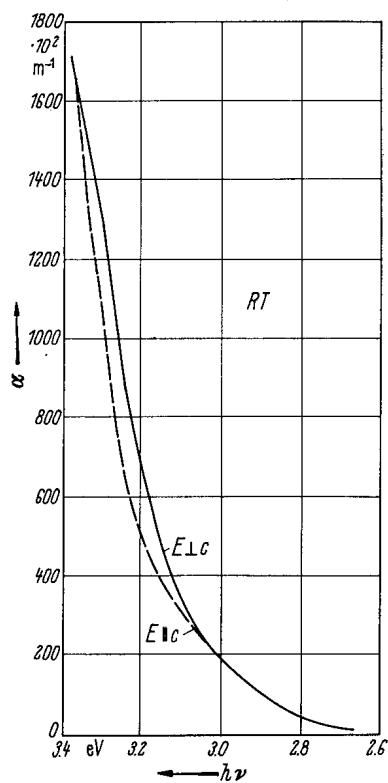
**Fig. 1A-10-111.** BaTiO<sub>3</sub>.  $R$ ,  $T$  vs.  $\nu$  [64Bal].  $R$ : reflectivity.  $T$ : transmission. Note change of scale at  $\nu = 4.2 \cdot 10^{12}$  Hz in (b).



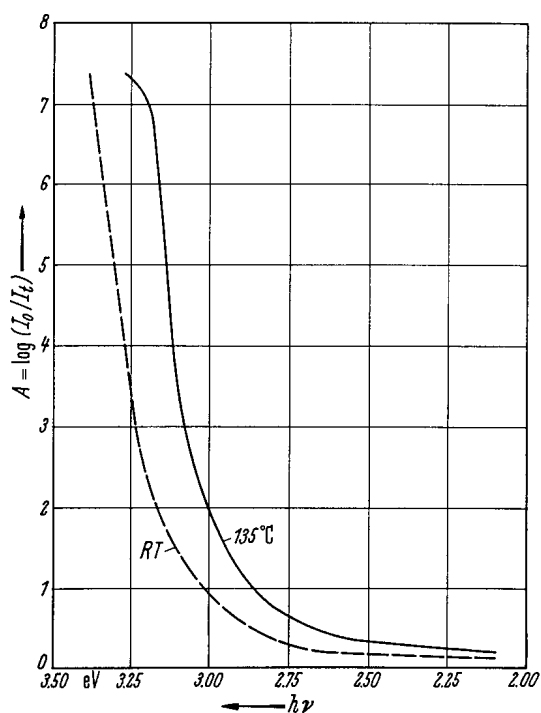
**Fig. 1A-10-112.**  $BaTiO_3$ .  $T$  vs.  $\lambda$  [64Ike].  $T$ : transmission for the infrared radiation.



**Fig. 1A-10-113.** BaTiO<sub>3</sub> (Fe doped). Infrared absorption spectra due to OH vibration [89Jov]. *A*: optical density ( $A = \log_{10} I_0/I_i$ ;  $I_0$ : intensity of incident light,  $I_i$ : intensity of transmitted light). Parameter: Fe concentration.  $T = \text{RT}$ .

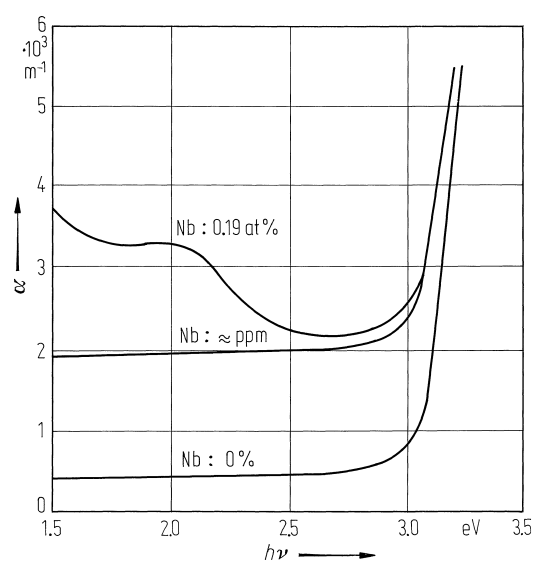


**Fig. 1A-10-114.**  $\text{BaTiO}_3$ .  $\alpha$  vs.  $h\nu$  [59Cas].  $\alpha$ : optical absorption coefficient. Polarized light.

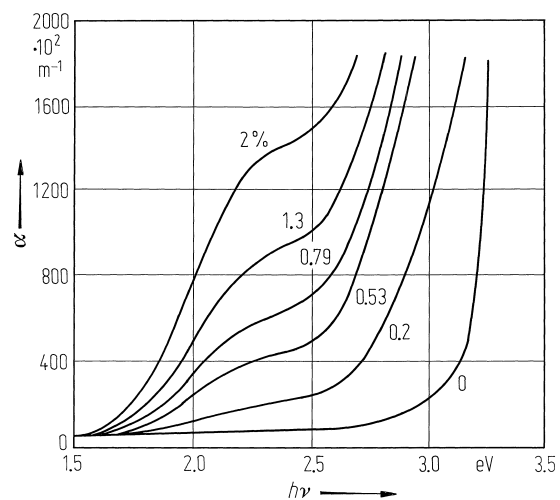


**Fig. 1A-10-115.** BaTiO<sub>3</sub>.  $A$  vs.  $h\nu$  [59Cas].  $A$ : optical density ( $A = \log_{10} I_0/I_t$ ;  $I_0$ : intensity of incident light,  $I_t$ : intensity of transmitted light). Unpolarized light.

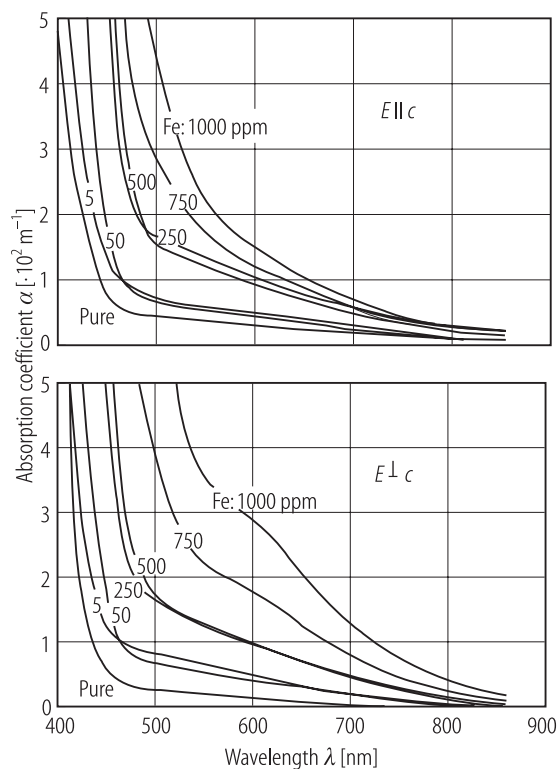




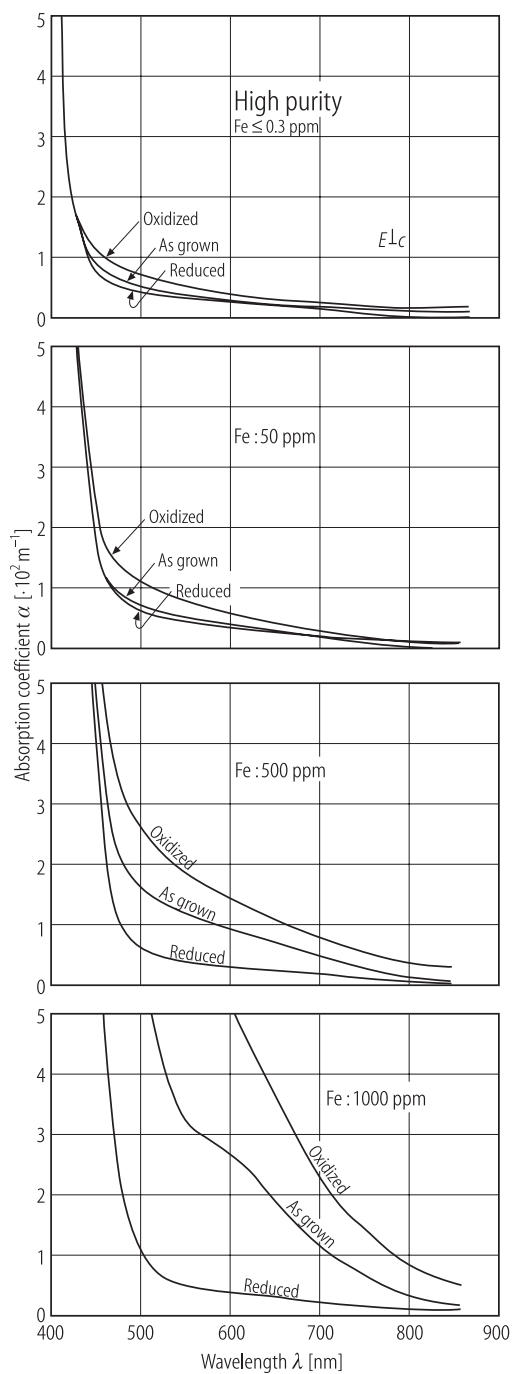
**Fig. 1A-10-116.** BaTiO<sub>3</sub> (Nb doped).  $\alpha$  vs.  $h\nu$  at RT [81Mor].  $\alpha$ : optical absorption coefficient. Parameter: Nb concentration. The crystal was grown by a pulling method.



**Fig. 1A-10-117.** BaTiO<sub>3</sub> (Co doped).  $\alpha$  vs.  $h\nu$  at RT [83Mic].  $\alpha$ : optical absorption coefficient. Parameter: Co concentration. The crystal was grown by the KF flux method.



**Fig. 1A-10-118.** BaTiO<sub>3</sub> (Fe doped).  $\alpha$  vs.  $\lambda$  at RT [88Sch2].  $\alpha$ : optical absorption coefficient. Polarized light:  $E \perp c$  and  $E \parallel c$ . Parameter: Fe concentration (with respect to Fe/Ti ratio in the melt). The crystal was grown by the top-seeded solution growth method.



**Fig. 1A-10-119.** BaTiO<sub>3</sub> (Fe doped).  $\alpha$  vs.  $\lambda$  at RT [88Sch2].  $\alpha$ : optical absorption coefficient. Polarized light:  $E$  perpendicular to the  $c$ -axis. Parameter: Fe concentration (with respect to Fe/Ti ratio in the melt). The crystal was grown by the top-seeded solution growth method. Oxidation was done at oxygen partial pressure  $p_{\text{O}_2} = 10^5 \text{ Pa}$  and 800 °C. Reduction was done at  $p_{\text{O}_2} = 10 \text{ Pa}$  and 800 °C.