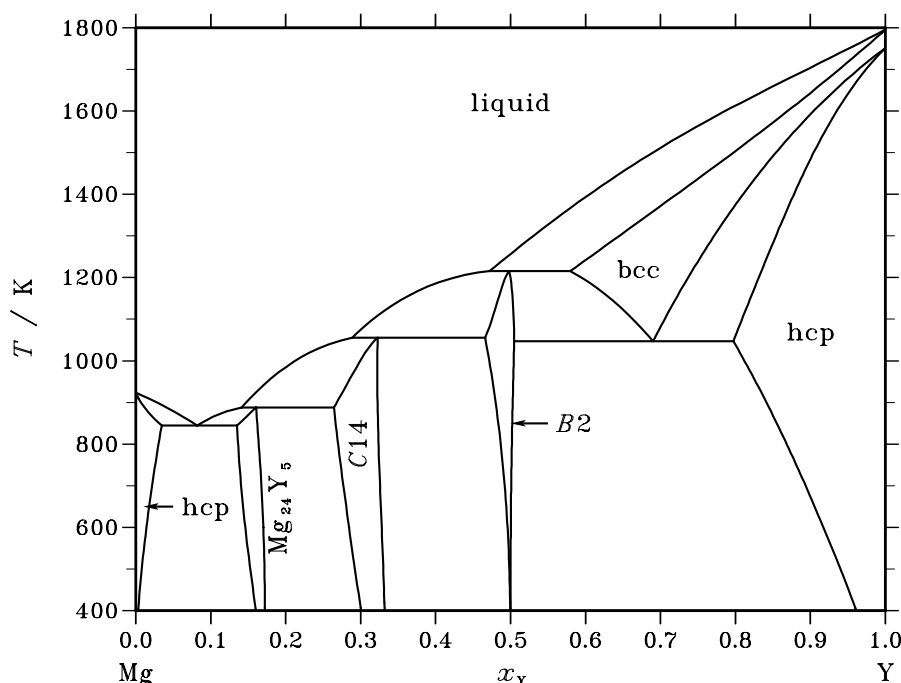


**Mg – Y (Magnesium – Yttrium)****Fig. 1.** Calculated phase diagram for the system Mg-Y.

Magnesium alloys are becoming increasingly important due to potential weight saving in comparison with aluminium based alloys. Yttrium additives are of interest because they enhance high-temperature properties and improve casting characteristics. Mg-Y alloys show higher creep resistance, better corrosion resistance, a considerable age hardening response and good strength properties at room temperature as well as at high temperatures. The assessment of thermodynamic data has been performed by [88Ran, 98Luk, 03Fab]. They are based on phase diagram information and thermodynamic data from [65Smi]. The more recent measurements of the enthalpy of mixing of liquid Y in Mg [91Aga, 91Feu, 95Aga], the data of Mg activity in the liquid phase [97Gan] and the enthalpy of formation values of the intermetallic phases from [89Pya, 90Pya] were additionally used by [03Fab]. The new data of [96Bon, 97Fla] on site occupancy in intermetallic compounds were also taken into account by [03Fab]. Therefore, the description of [03Fab] is recommended here. The system Mg-Y is characterised by complete solubility in the liquid state and limited solubility of Mg in solid Y and vice versa. Three intermetallic compounds  $\text{MgY}_{1-x}$ ,  $\text{Mg}_2\text{Y}_{1-x}$  and  $\text{Mg}_{24}\text{Y}_{5-x}$  with limited homogeneity range exist in this system. The  $\text{MgY}_{1-x}$  phase is described as a highly ordered B2 phase originating from the disordered bcc-Y phase. The phases  $\text{Mg}_2\text{Y}_{1-x}$  and  $\text{Mg}_{24}\text{Y}_{5-x}$  are modelled in accordance with experimental site occupancy data [96Bon, 97Fla].

**Table I.** Phases, structures and models.

Phase	Strukturbericht	Prototype	Pearson symbol	Space group	SGTE name	Model
liquid					LIQUID	$(\text{Mg}, \text{Y})_1$
hcp	A3	Mg	<i>hP2</i>	<i>P6<sub>3</sub>/mmc</i>	HCP_A3	$(\text{Mg}, \text{Y})_1$
$\text{Mg}_{24}\text{Y}_5$	A12	$\alpha\text{Mn}$	<i>cI58</i>	<i>I43m</i>	MG24Y5	$\text{Mg}_{24}(\text{Mg}, \text{Y})_4\text{Y}_1$
C14	C14	$\text{MgZn}_2$	<i>hP12</i>	<i>P6<sub>3</sub>/mmc</i>	LAVES_C14	$(\text{Mg}, \text{Y})_2(\text{Mg}, \text{Y})_1$
B2	B2	CsCl	<i>cP2</i>	<i>Pm3m</i>	BCC_B2	$(\text{Mg}, \text{Y})_1(\text{Mg}, \text{Y})_1$
bcc	A2	W	<i>cI2</i>	<i>Im3m</i>	BCC_A2	$(\text{Mg}, \text{Y})_1$

**Table II.** Invariant reactions.

Reaction	Type	$T / \text{K}$	Compositions / $x_Y$			$\Delta_r H / (\text{J/mol})$
liquid + bcc $\rightleftharpoons B2$	peritectic	1215.2	0.472	0.580	0.498	–13794
liquid + $B2 \rightleftharpoons C14$	peritectic	1055.5	0.289	0.466	0.323	–8869
bcc $\rightleftharpoons B2 + \text{hcp}$	eutectoid	1047.2	0.690	0.505	0.797	–4142
liquid + $C14 \rightleftharpoons \text{Mg}_{24}\text{Y}_5$	peritectic	887.6	0.141	0.264	0.160	–7650
liquid $\rightleftharpoons \text{hcp} + \text{Mg}_{24}\text{Y}_5$	eutectic	844.9	0.082	0.035	0.135	–7733

**Table IIIa.** Integral quantities for the stable phases at 1223 K.

Phase	$x_Y$	$\Delta G_m$ [J/mol]	$\Delta H_m$ [J/mol]	$\Delta S_m$ [J/(mol·K)]	$G_m^E$ [J/mol]	$S_m^E$ [J/(mol·K)]	$\Delta C_P$ [J/(mol·K)]
liquid	0.000	0	0	0.000	0	0.000	0.000
	0.100	–5346	–3555	1.464	–2041	–1.238	0.606
	0.200	–8298	–5533	2.261	–3209	–1.900	1.211
	0.300	–9837	–6121	3.039	–3625	–2.040	1.817
	0.400	–10252	–5508	3.879	–3408	–1.717	2.422
	0.477	–9922	–4336	4.568	–2884	–1.187	2.889
bcc	0.586	–9096	–9757	–0.540	–2198	–6.181	–1.349
	0.600	–8979	–9426	–0.365	–2135	–5.961	–1.379
	0.700	–7731	–6632	0.899	–1519	–4.180	–1.588
	0.745	–6929	–5117	1.482	–1151	–3.242	–1.681
hcp	0.836	–5122	–4802	0.261	–589	–3.445	–0.020
	0.900	–3682	–3058	0.510	–376	–2.193	–0.012
	1.000	0	0	0.000	0	0.000	0.000

Reference states: Mg(liquid), Y(hcp)

**Table IIIb.** Partial quantities for Mg in the stable phases at 1223 K.

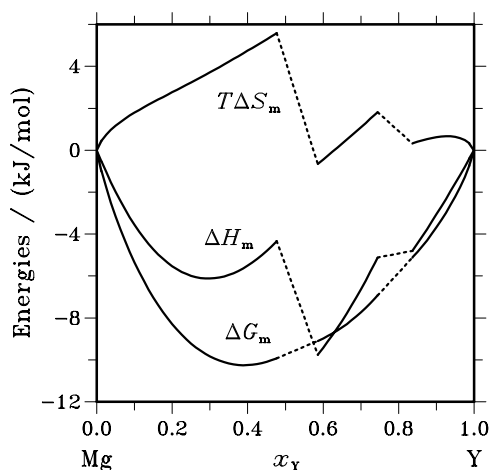
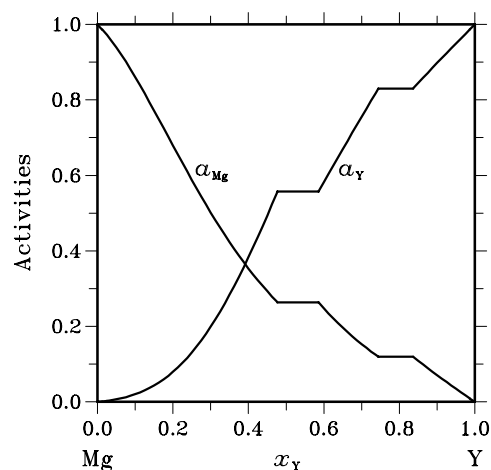
Phase	$x_{\text{Mg}}$	$\Delta G_{\text{Mg}}$ [J/mol]	$\Delta H_{\text{Mg}}$ [J/mol]	$\Delta S_{\text{Mg}}$ [J/(mol·K)]	$G_{\text{Mg}}^E$ [J/mol]	$S_{\text{Mg}}^E$ [J/(mol·K)]	$a_{\text{Mg}}$	$\gamma_{\text{Mg}}$
liquid	1.000	0	0	0.000	0	0.000	1.000	1.000
	0.900	–1528	–821	0.578	–456	–0.298	0.861	0.956
	0.800	–3934	–3031	0.739	–1665	–1.117	0.679	0.849
	0.700	–7014	–6253	0.623	–3387	–2.343	0.502	0.717
	0.600	–10578	–10109	0.383	–5383	–3.864	0.353	0.589
	0.523	–13549	–13275	0.224	–6957	–5.166	0.264	0.505
bcc	0.414	–13549	–22847	–7.603	–4591	–14.928	0.264	0.637
	0.400	–14199	–23604	–7.690	–4882	–15.308	0.247	0.619
	0.300	–19116	–29141	–8.197	–6874	–18.207	0.153	0.509
	0.255	–21603	–31748	–8.296	–7724	–19.644	0.119	0.468
hcp	0.164	–21603	–26601	–4.087	–3201	–19.133	0.119	0.730
	0.100	–26979	–28851	–1.531	–3565	–20.676	0.070	0.704
	0.000	– $\infty$	–32450	$\infty$	–3923	–23.326	0.000	0.680

Reference state: Mg(liquid)

**Table IIIc.** Partial quantities for Y in the stable phases at 1223 K.

Phase	$x_Y$	$\Delta G_Y$ [J/mol]	$\Delta H_Y$ [J/mol]	$\Delta S_Y$ [J/(mol·K)]	$G_Y^E$ [J/mol]	$S_Y^E$ [J/(mol·K)]	$a_Y$	$\gamma_Y$
liquid	0.000	$-\infty$	-44075	$\infty$	-25169	-15.459	0.000	0.084
	0.100	-39717	-28170	9.442	-16303	-9.703	0.020	0.201
	0.200	-25753	-15541	8.349	-9387	-5.032	0.079	0.397
	0.300	-16423	-5812	8.676	-4180	-1.334	0.199	0.663
	0.400	-9762	1395	9.123	-445	1.504	0.383	0.957
	0.477	-5946	5465	9.330	1581	3.176	0.557	1.168
bcc	0.586	-5946	-494	4.458	-505	0.009	0.557	0.952
	0.600	-5499	26	4.517	-304	0.270	0.582	0.971
	0.700	-2851	3015	4.797	776	1.831	0.755	1.079
	0.745	-1896	4018	4.836	1103	2.384	0.830	1.115
hcp	0.836	-1896	-535	1.113	-78	-0.374	0.830	0.992
	0.900	-1093	-192	0.737	-22	-0.139	0.898	0.998
	1.000	0	0	0.000	0	0.000	1.000	1.000

Reference state: Y(hcp)

**Fig. 2.** Integral quantities of the stable phases at  $T=1223$  K.**Fig. 3.** Activities in the stable phases at  $T=1223$  K.**Table IV.** Standard reaction quantities at 298.15 K for the compounds per mole of atoms.

Compound	$x_Y$	$\Delta_f G^\circ$ / (J/mol)	$\Delta_f H^\circ$ / (J/mol)	$\Delta_f S^\circ$ / (J/(mol·K))	$\Delta_f C_P^\circ$ / (J/(mol·K))
Mg <sub>24</sub> Si <sub>5</sub>	0.170	-7368	-7707	-1.137	0.000
C14	0.320	-11645	-12165	-1.743	0.000
B2	0.500	-14473	-15580	-3.716	0.023

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